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**INSTYTUT METALURGII I INŻYNIERII MATERIAŁOWEJ**  
**im. Aleksandra Krupkowskiego**  
**Polskiej Akademii Nauk**

# **Photovoltaic systems – theory and practice**

**Part 4**

**Marek Lipiński**

**Kraków 2020**

*Projekt nr WND-POWR.03.02.00-00-IO43/16*

*Międzynarodowe interdyscyplinarne studia doktoranckie z zakresu nauk o materiałach z wykładowym językiem angielskim  
Program Operacyjny Wiedza Edukacja Rozwój 2014-2020, Działanie 3.2 Studia doktoranckie*



# Cours description

## 1. Introduction to photovoltaics

*Basic information about the solar energy and photovoltaic Energy conversion*

## 2. Technology of solar cells

*The industrial technology of silicon solar cells and thin films solar cells will be presented*

## 3. Emerging photovoltaics

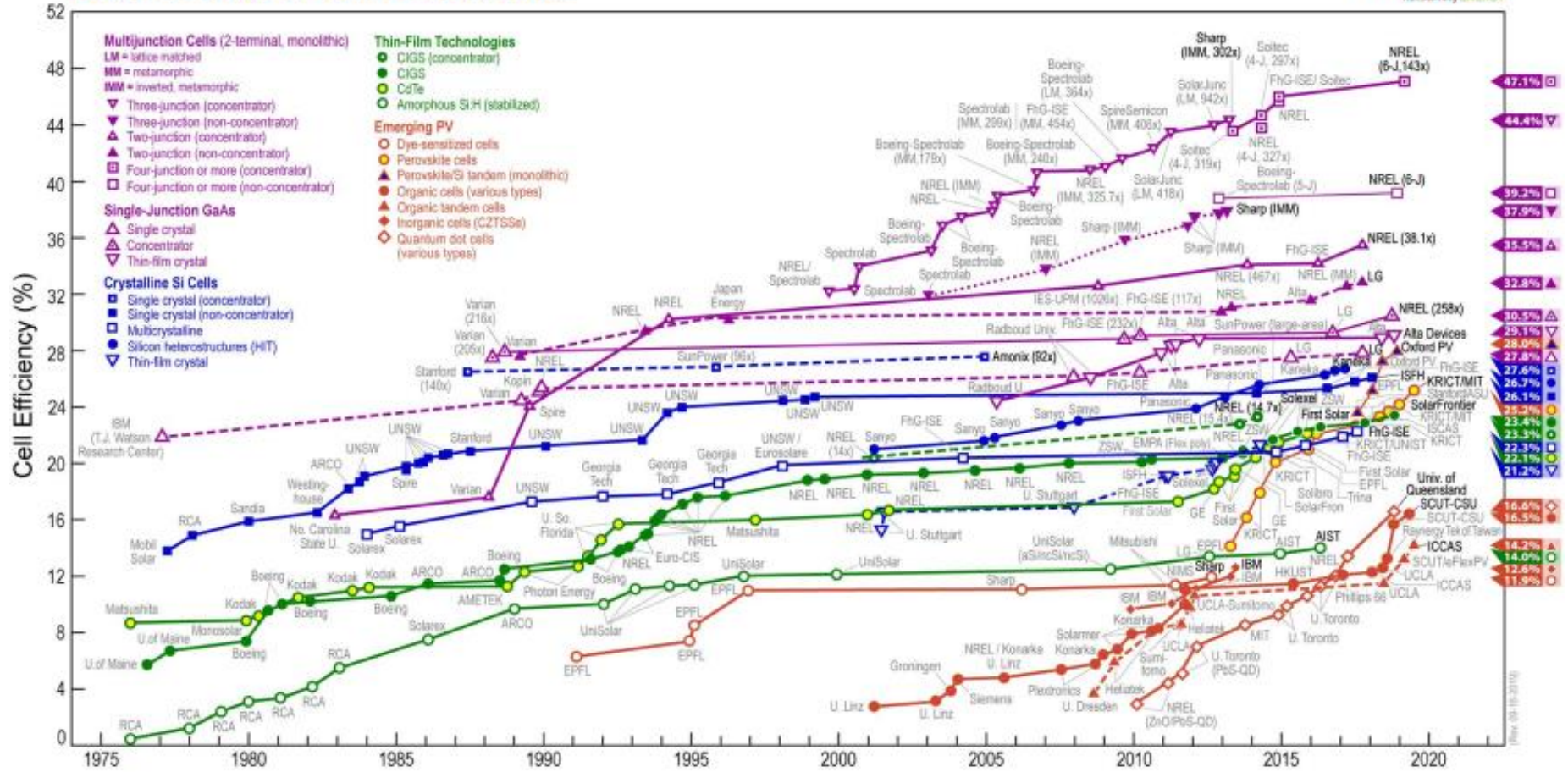
*Emerging materials and devices including dye-sensitized solar cell, organic solar cell, perovskite solar cell and quantum dot solar cell*

## 4. Photovoltaic systems

*Technology, applications, economics of photovoltaic systems*

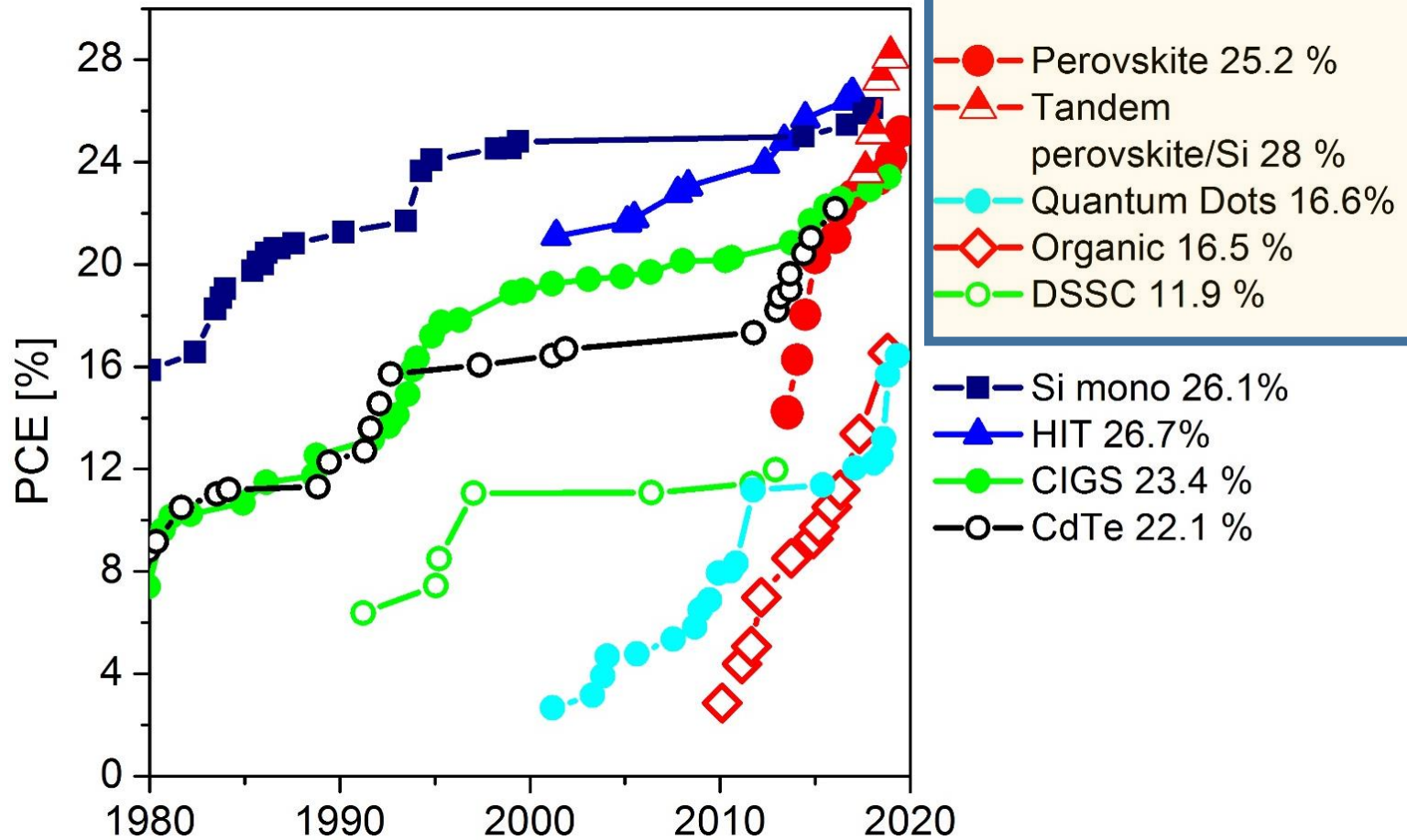


# Best Research-Cell Efficiencies





## Emerging PV





# QD solar cells



In a semiconductor crystallite whose size is smaller than twice the size of its exciton Bohr radius ( $a_b$ ), the excitons are squeezed, leading to quantum confinement.



## Material properties and Bohr radii $a_B$ of various bulk semiconductors

		$E_g$ (eV)	$m_c^*/m_0$	$m_h^*/m_0$	Electron $a_B$ (nm)		Hole $a_B$ (nm)	Exciton $a_B$ (nm)
II-VI	CdS	2.48	0.25	0.6	5	1	<1	2
	CdSe	1.73	0.12	0.9 <sup>a</sup>	6	3	1 <sup>a</sup>	4
	CdTe	1.48	0.09	0.8 <sup>a</sup>	7	4	1 <sup>a</sup>	5
III-V	InP	1.34	0.073	0.45 <sup>a</sup>	11	7	1	8
	InAs	0.35	0.023	0.57 <sup>a</sup>	12	27	2	29
	InSb	0.17	0.012	0.44 <sup>a</sup>	16	59	2	61
IV-VI	PbS <sup>c</sup>	0.42	0.087 <sup>b</sup>	0.083 <sup>b</sup>	17	10	11	21
	PbSe <sup>c</sup>	0.28	0.047 <sup>b</sup>	0.041 <sup>b</sup>	23	26	29	55
	PbTe <sup>c</sup>	0.31	0.034 <sup>b</sup>	0.032 <sup>b</sup>	33	56	48	104

$$a_B = \epsilon m_0 / m^* a_0$$

$a_0$  Bohr radius  $a_0 \approx 0.53$  nm

$\epsilon$  semiconductor dielectric constant

$m^*$  effective mass of electron or hole

or for exciton reduced mass ( $m_{\text{exe}}^{*-1} = m_e^{*-1} + m_h^{*-1}$ )

[1] O. Madelung, Semiconductors: Data Handbook. Berlin:Springer-Verlag,2004.

[2] T. D. Krauss and J. J. Peterson in Colloidal Quantum Dot Optoelectronics and Photovoltaics, edited by G. Konstantatos and E. H. Sargent



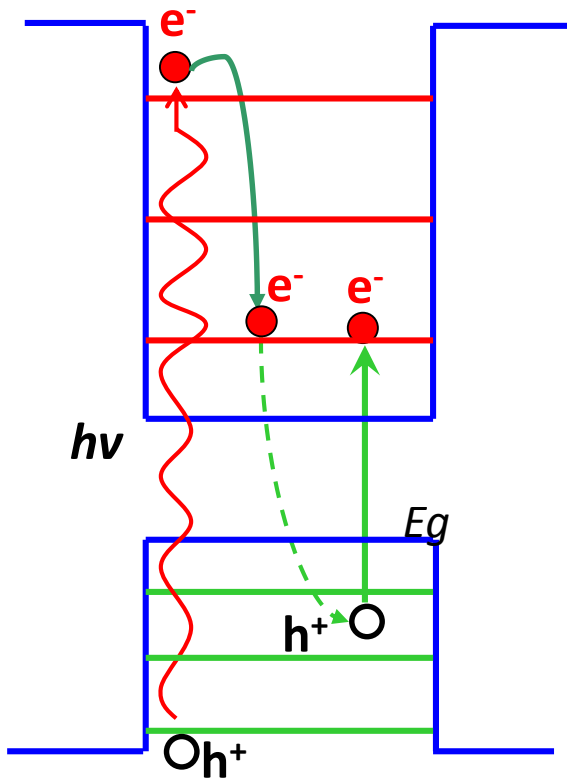
# QD solar cells

Semiconductor	Size (nm)	Eg (eV)
PbSe	$\infty$	0.27 i
PbSe	5.4	0.73
PbSe	4.7	0.82
PbSe	3.9	0.91
PbS	$\infty$	0.41 i
PbS	5.5	0.85
PbTe	$\infty$	0.31 i
PbTe	5.5	0.91
Si	$\infty$	1.12
Si	2	1.7





# QD solar cells



One photon two pairs e-h

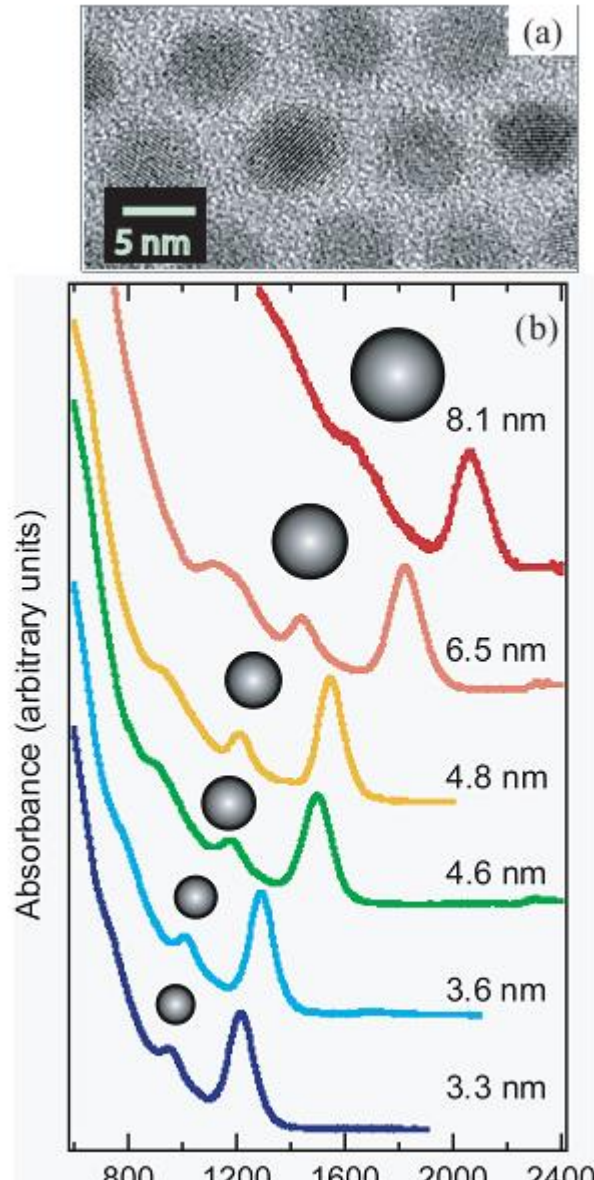
## The effects of quantum confinement in quantum dots

- Slowing down the cooling of excitons, strengthening the Auger process,
- There is no request to maintain the crystalline momentum
- One photon can generate many e-h pairs. Crash ionization multi-exciton generation by high energy photons
- Expanding the energy gap

- A.J. Nozik / Physica E 14 (2002) 115
- A. Luque, A. Marti, and A.J. Nozik, MRS Bulletin Vol. 32 (2007) 236



# QD solar cells

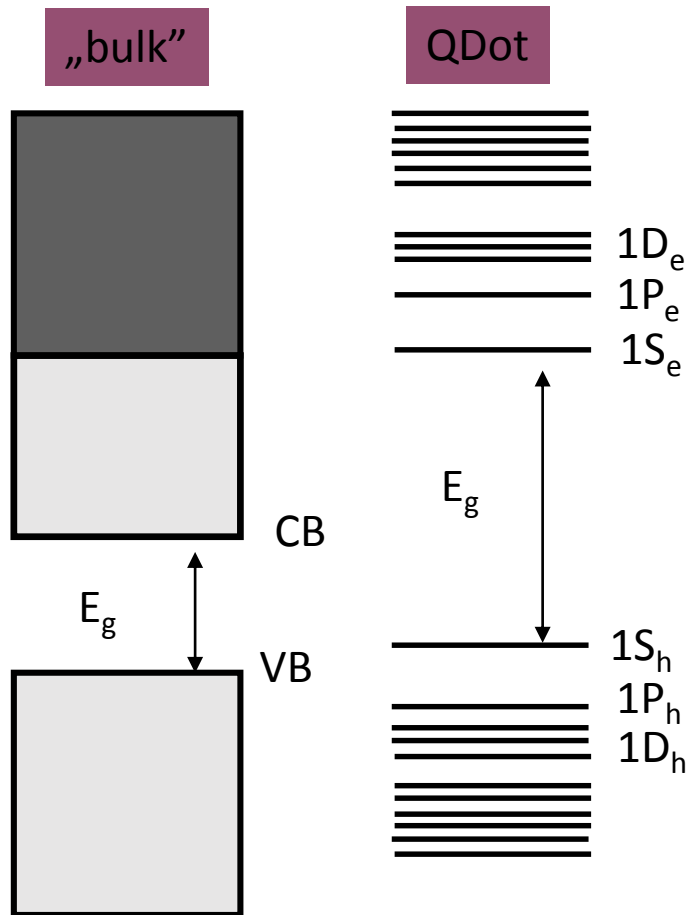


(a) TEM image showing PbSe NCs with average diameter of 5.2nm. (b) Linear absorption spectra of a series of PbSe NCs with average diameter ranging from 3.3nm to 8.1nm. Strong excitonic absorption and a blue-shift of the onset are signatures of quantum confinement in NCs.

<https://onlinelibrary.wiley.com/doi/pdf/10.1002/lpor.200810013>



# QD solar cells



Quantum dot fluorescence (Q-dots)  
CdSe of various sizes when exposed to  
UV light



# QD solar cells

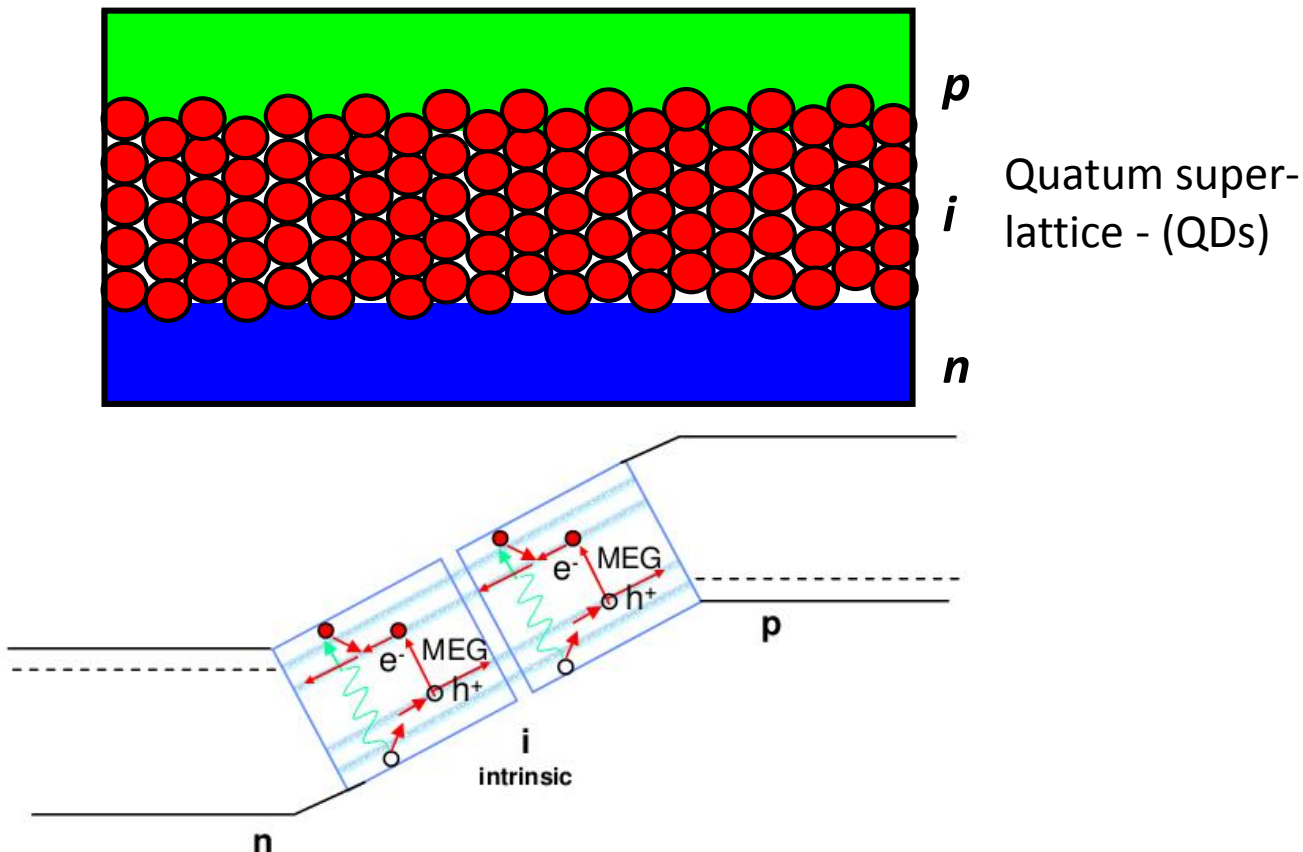
## *Solar cell configurations with quantum dots:*

1. Photoelectrodes from QDs "arrays,, (super-lattice)
2. Nanocrystalline  $\text{TiO}_2$  sensitized QDs
3. QDs immersed in a polymer blend ("e" and "h")



# QD solar cells

## 1. Quantum super-lattice – p-i-n structure.

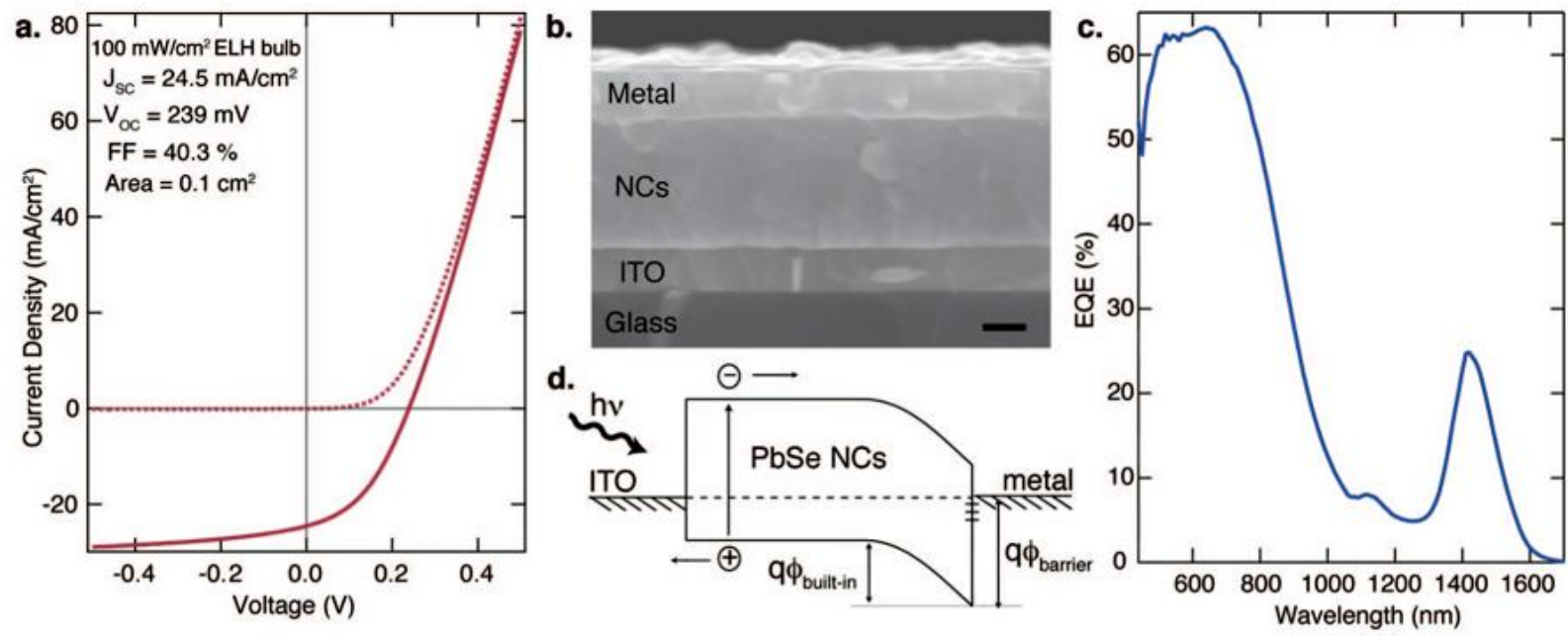


Slow down cooling process, transport and collection of hot carriers in p and n contacts - higher voltage or larger photocurrent as a result of MEG



# QD solar cells

## Quantum super-lattice– Structure with a Schottki diode

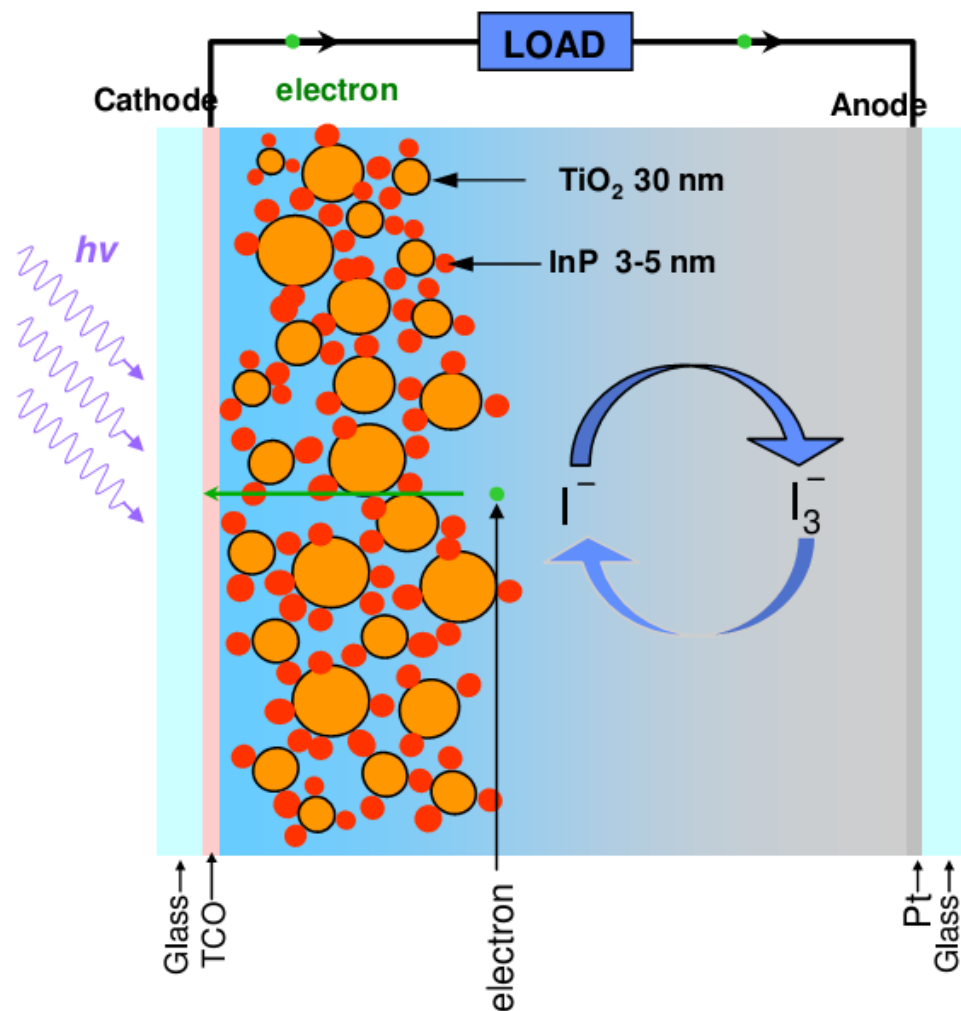


NCs layer– deposited by „layer – by layer (LBL) dip coating”, 60 nm thickness

J. M. Luther, M. Law, M. C. Beard, Q., M.O. Reese, R. J. Ellingson, and A. J. Nozik, Nano Lett., Vol. 8, No. 10, 2008, p.3488.



## 2. QDs DSSC



1. Electrons of QD are excited by solar energy adsorption
2. Electron transfer from QD to TCO via TiO<sub>2</sub>
3. Electrons get to the counter electrode after working at external load
4.  $\frac{1}{2} I_3^- - e^- \rightarrow \frac{3}{2} I^-$  at counter electrode
5.  $\frac{3}{2} I^- \rightarrow \frac{1}{2} I_3^- + e^-$  at QD

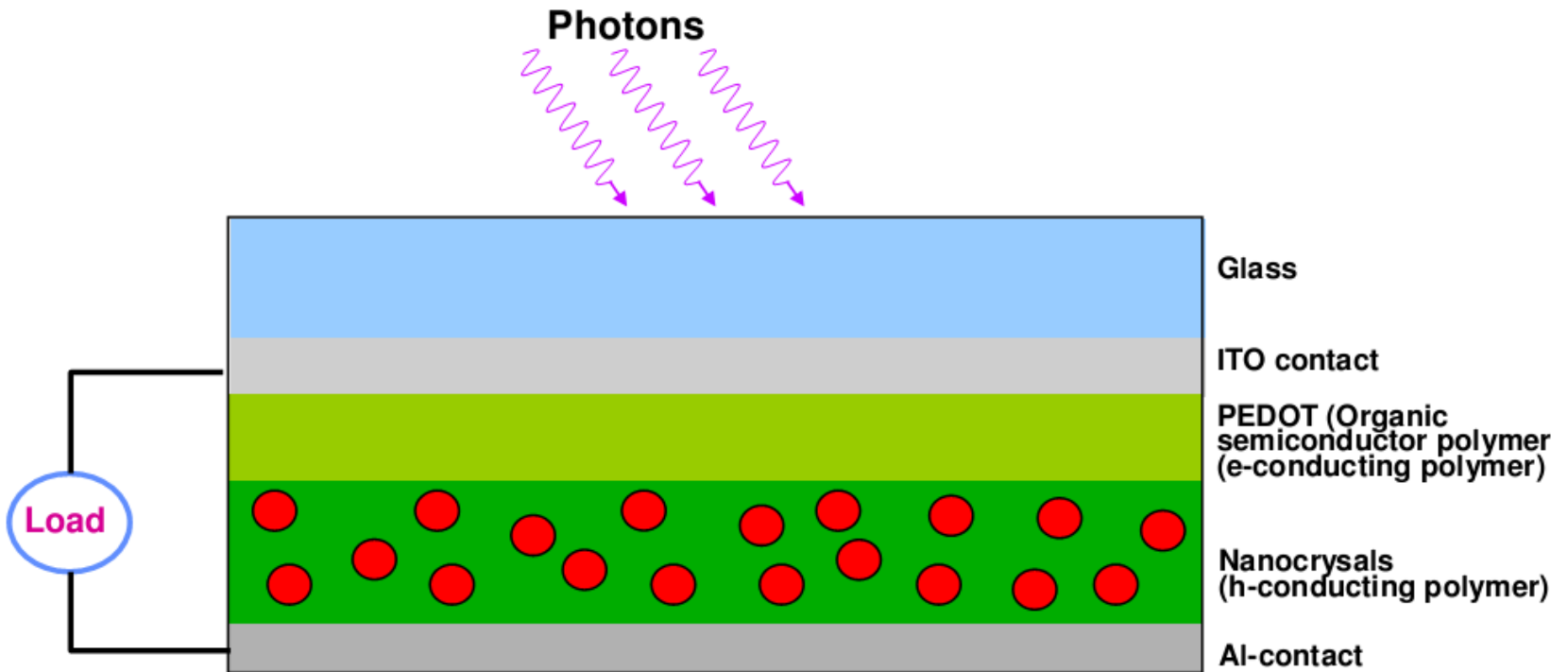
TiO<sub>2</sub> : QD-sensitized negative charge carrier  
Electrolyte: iodine-triiodide redox couple  
QD: InP, CdSe, CdTe, PbS, etc.  
TCO: Transparent conducting oxide  
Pt : Catalyst

A. Luque, A. Marti, and J. Nozik, MRS Bulletin Vol. 32 (2007) p. 236

Barwnik w ogniwach DSSC zastąpiony jest przez QDs.



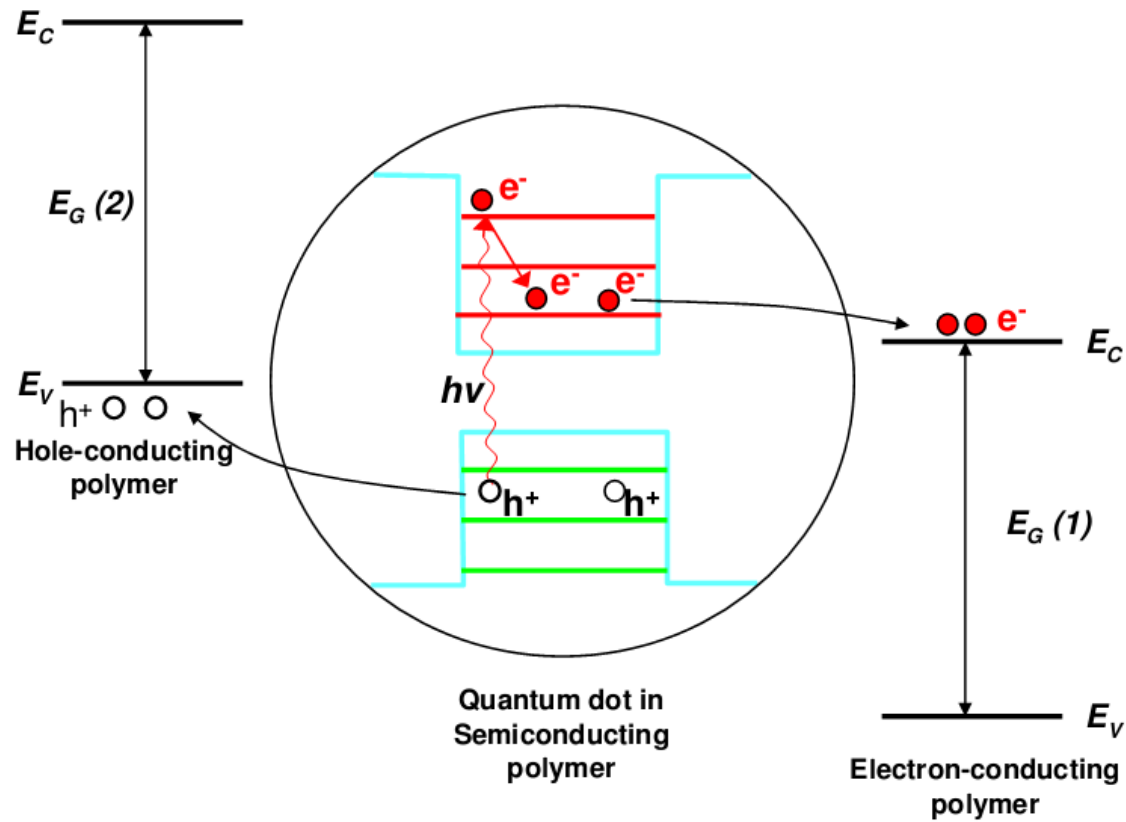
# QDs immersed in a polymer blend





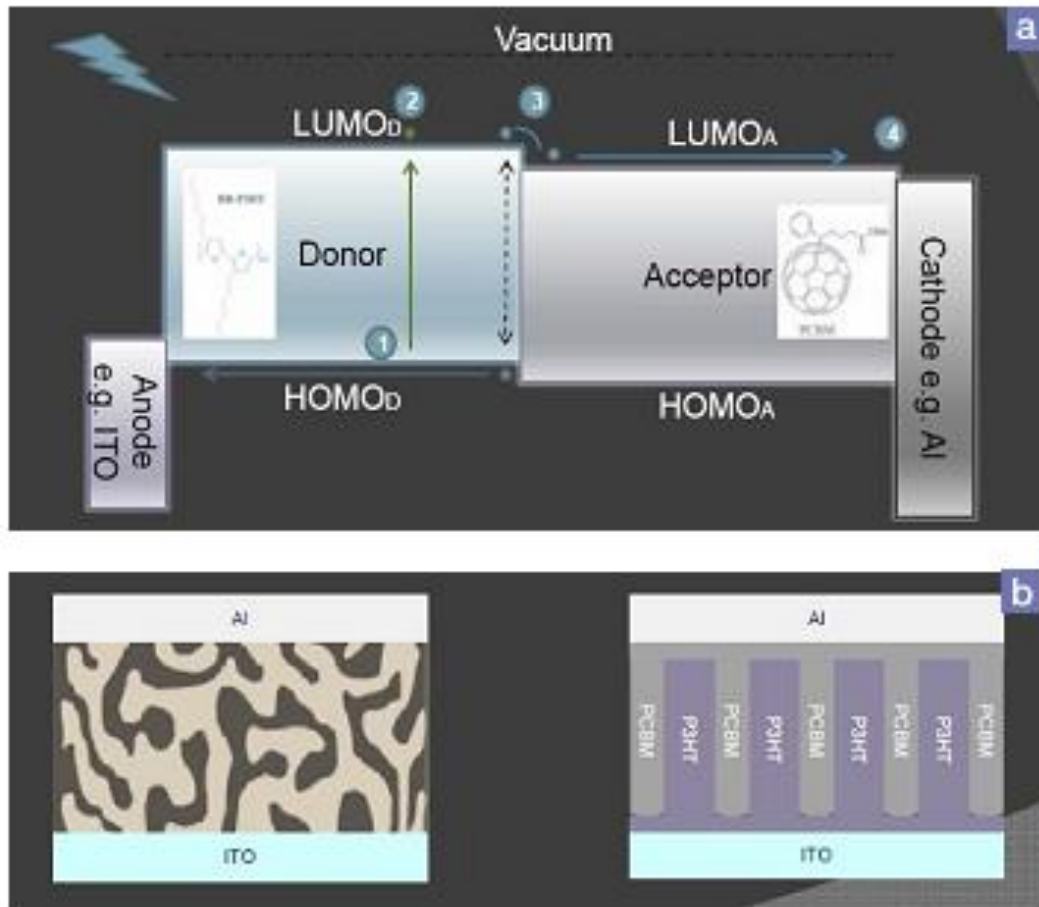


# QDs immersed in a polymer blend





# QDs immersed in a polymer blend



The schematic band energy structure of bulk-heterojunction organic cell used to produce the results (b)(left) a schematic showing random distribution of donor (P3HT) and acceptor (PCBM) regions in the blended bulk-heterojunction organic solar cell, (right) a schematic diagram of systematic alignment of donor and acceptor layers



# QDs immersed in a polymer blend

Materials Views

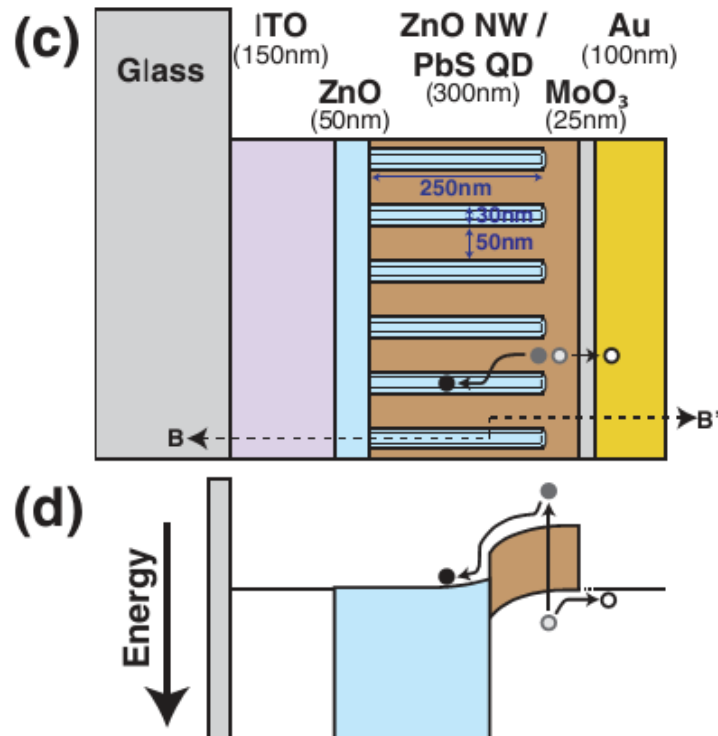
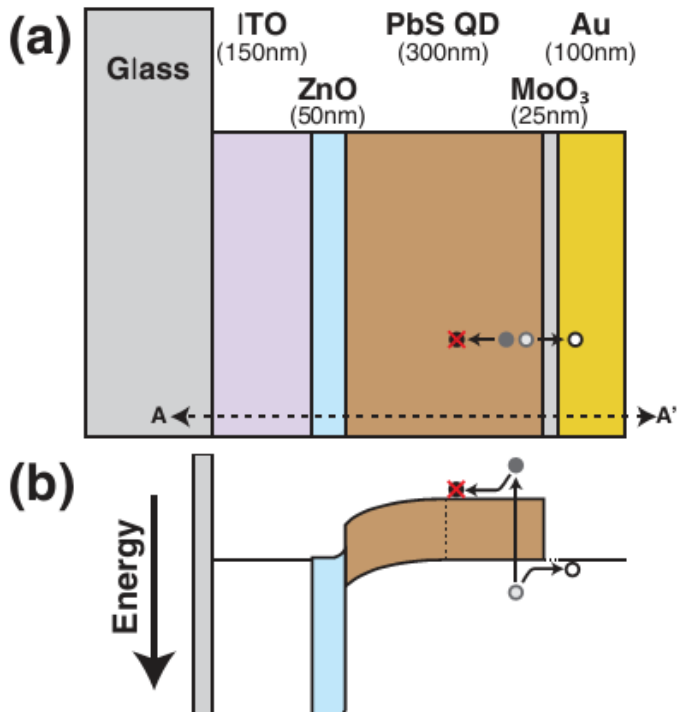
www.MaterialsViews.com

ADVANCED MATERIALS

www.advmat.de

## ZnO Nanowire Arrays for Enhanced Photocurrent in PbS Quantum Dot Solar Cells

Joel Jean, Sehoon Chang, Patrick R. Brown, Jayce J. Cheng, Paul H. Rekemeyer, Mounji G. Bawendi, Silvija Gradečak, and Vladimir Bulović\*



**Figure 1.** (a) Schematic and (b) energy band diagram at short-circuit (cross-section along A-A') of a planar QDPV device with parallel light absorption and carrier collection pathways. By incorporating solution-processed ZnO nanowires, an ordered bulk heterojunction (BHJ) architecture – shown here by: (c) schematic and (d) energy band diagram (cross-section along B-B') – can decouple absorption from collection, extending the effective depletion width throughout a thick OD film.



# IBSC cells from colloidal solutions of quantum dots

Nature Communications (2019)

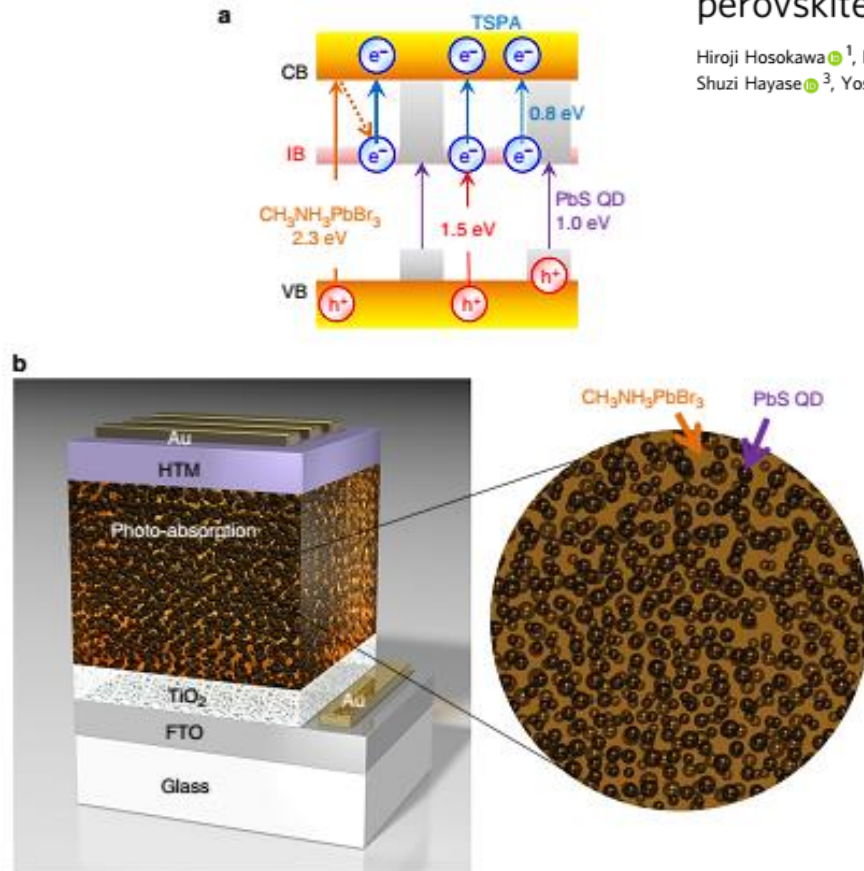
ARTICLE

DOI: 10.1038/s41467-018-07655-3

OPEN

## Solution-processed intermediate-band solar cells with lead sulfide quantum dots and lead halide perovskites

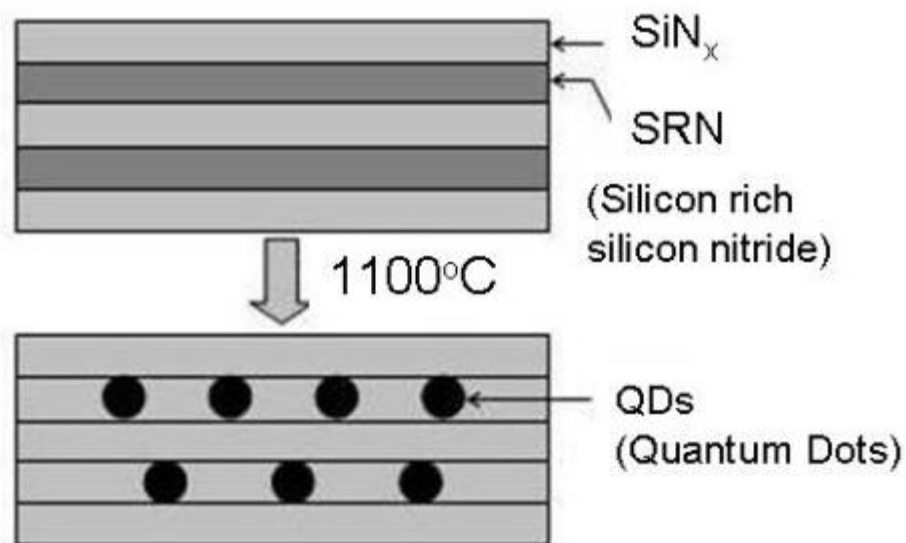
Hiroji Hosokawa<sup>1</sup>, Ryo Tamaki<sup>2</sup>, Takuya Sawada<sup>1</sup>, Akinori Okonogi<sup>1</sup>, Haruyuki Sato<sup>1</sup>, Yuhei Ogomi<sup>3</sup>, Shuzi Hayase<sup>3</sup>, Yoshitaka Okada<sup>2</sup> & Toshihiro Yano<sup>1</sup>



IBSC made from solutions. PbS quantum dots (4 nm) immersed in MAPbBr<sub>3</sub> perovskite



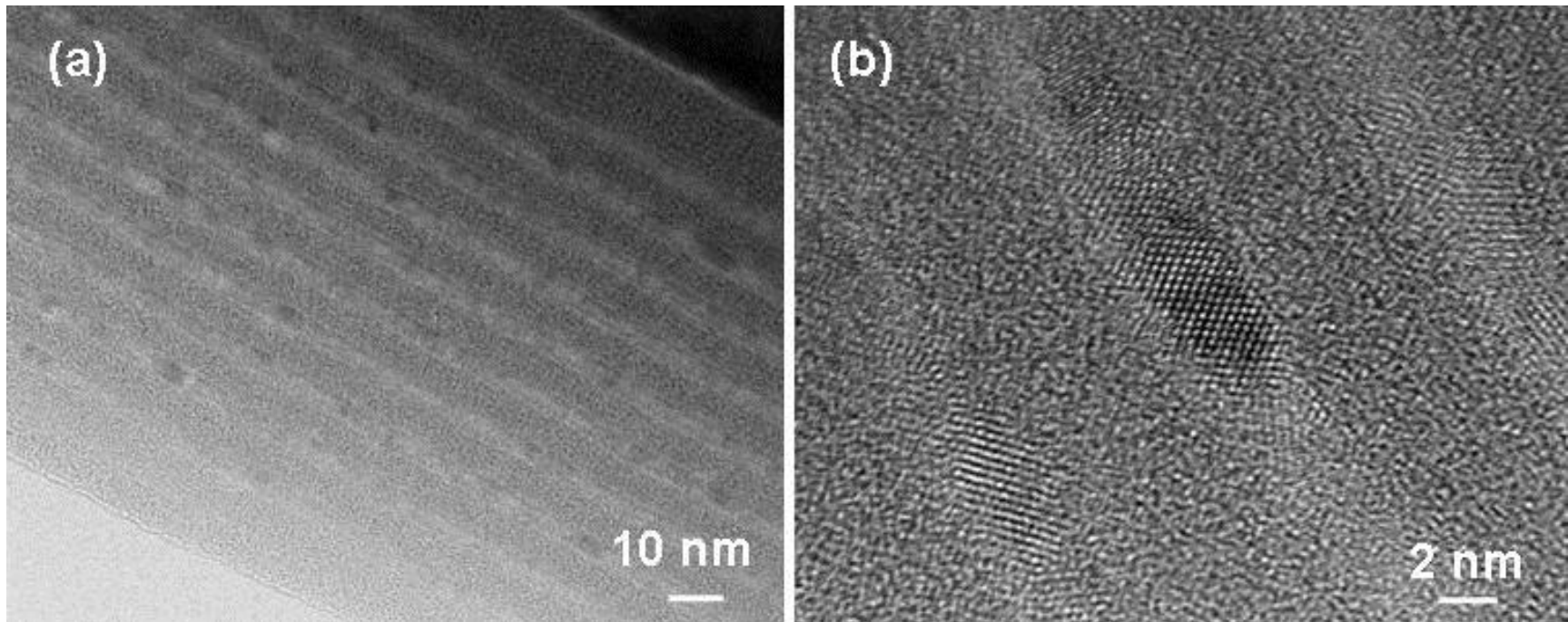
# Silicon Quantum dots for III generation solar cells



**Fig. 1.** Scheme of formation of quantum dots (QDs) in silicon nitride multilayer according to Zacharias and Green (M. Lipiński in *Archives of Materials Science and Engineering*, 46 (2010) 69-87).



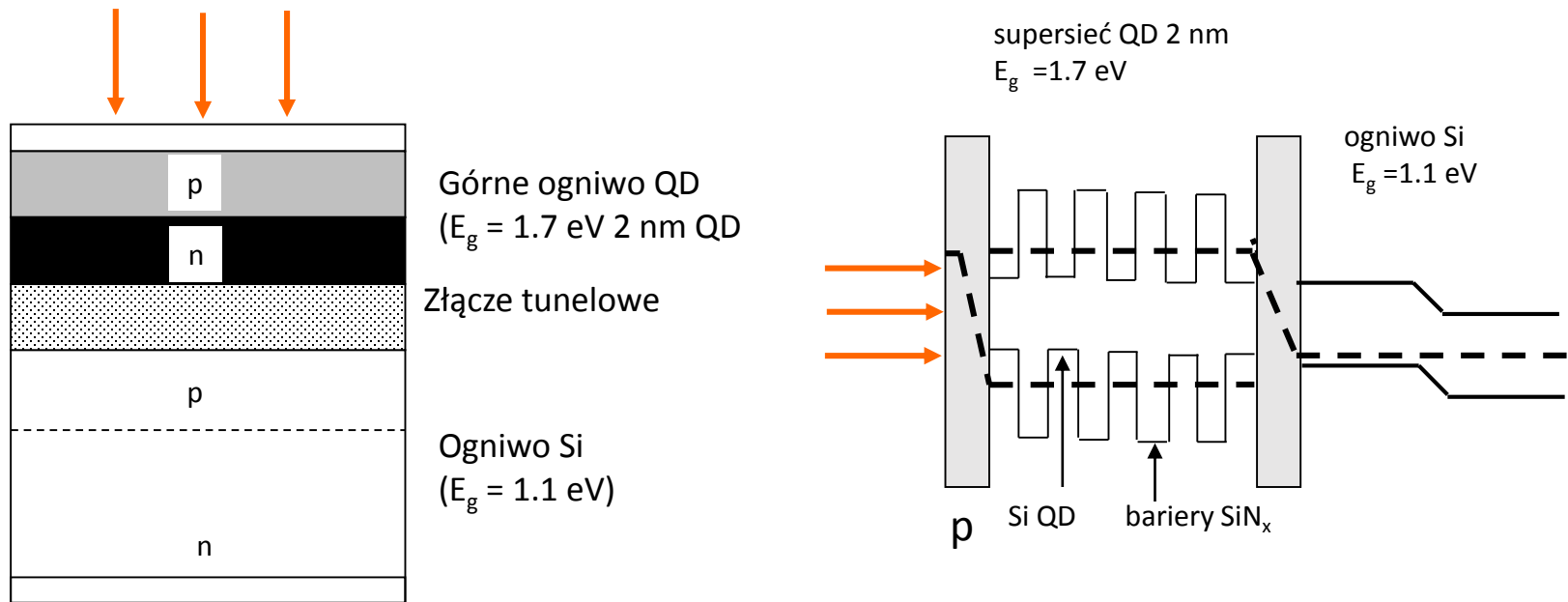
# Silicon Quantum dots for III generation solar cells



Cross-sectional TEM images of the multilayer (a) low and (b) high magnification. The sample was annealed at 1100°C (M. Lipiński in Archives of Materials Science and Engineering, 46 (2010) 69-87).



# Tandem silicon solar cells using Si Qdots



a) Scheme of a two-junction silicon cell. The upper cell is made of a silicon superlattice quantum dots with an energy gap of 1.7 eV, the bottom cell is a classic silicon link. These cells are connected with each other by a tunnel junction [1].

b) Scheme of the band structure of a two-junction cell [2].

[1] E.-C. Cho, M.A.Green, G. Conibeer et al., *Silicon quantum dots in a dielectric matrix for all-silicon tandem solar cells*, Hindwai Publishing Corporation, *Advances in OptoElectronics* (2007) 1-11

2] G. Coniber, *Third-generation photovoltaics*, *Materials Today* 10 (2007) 42-50.



# Light converter from perovskite nanoparticles

## STUDYING OF PEROVSKITE NANOPARTICLES IN PMMA MATRIX USED AS LIGHT CONVERTER FOR SILICON SOLAR CELL

Arch. Metall. Mater. **62** (2017), 3, 17331-1739

M. LIPIŃSKI<sup>#</sup>, R.P. SOCHA<sup>\*\*</sup>, A. KĘDRA<sup>\*\*</sup>, K. GAWLIŃSKA<sup>\*</sup>, G. KULESZA-MATLAK<sup>\*</sup>,  
Ł. MAJOR<sup>\*</sup>, K. DRABCZYK<sup>\*</sup>, K. ŁABA<sup>\*\*\*</sup>, Z. STAROWICZ<sup>\*</sup>, K. GWÓZDŹ<sup>\*\*\*\*</sup>, A. GÓRAL<sup>\*</sup>, E. POPKO<sup>\*\*\*\*</sup>

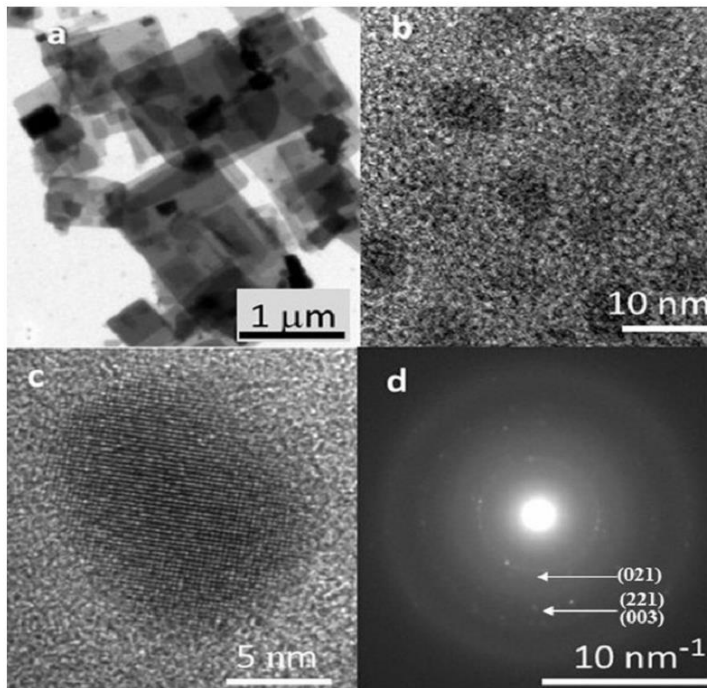
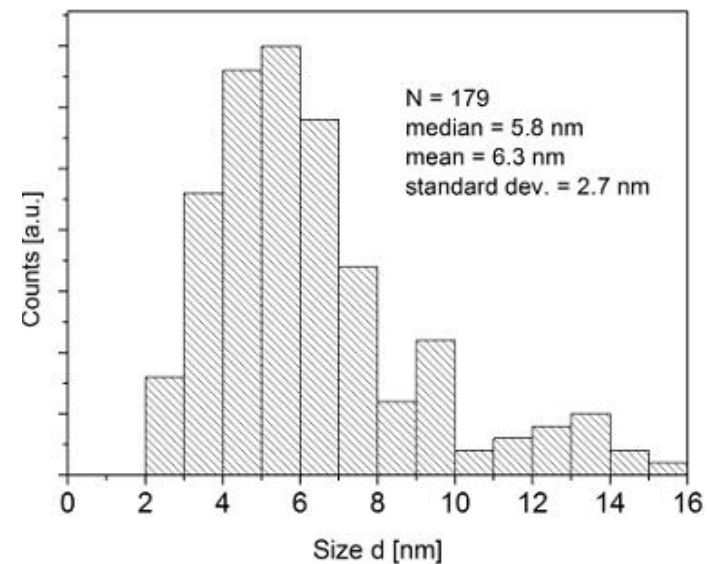


Fig. 1. The TEM image of the  $\text{CH}_3\text{NH}_3\text{PbBr}_3$  perovskite morphology (a), High Resolution TEM images (b,c) and electron diffraction pattern with two rings corresponded to (021), (221) and (003) faces of cubic phase according to ref. [27] (d) of the nanoparticles

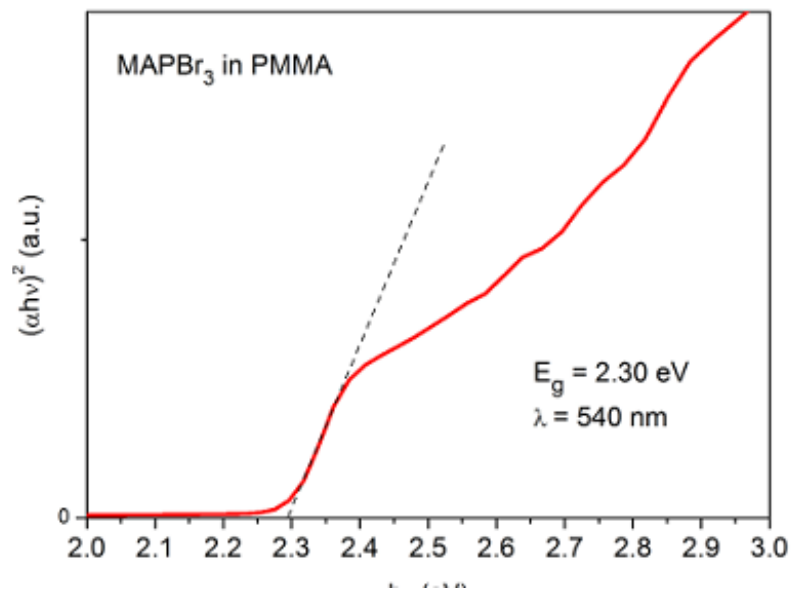
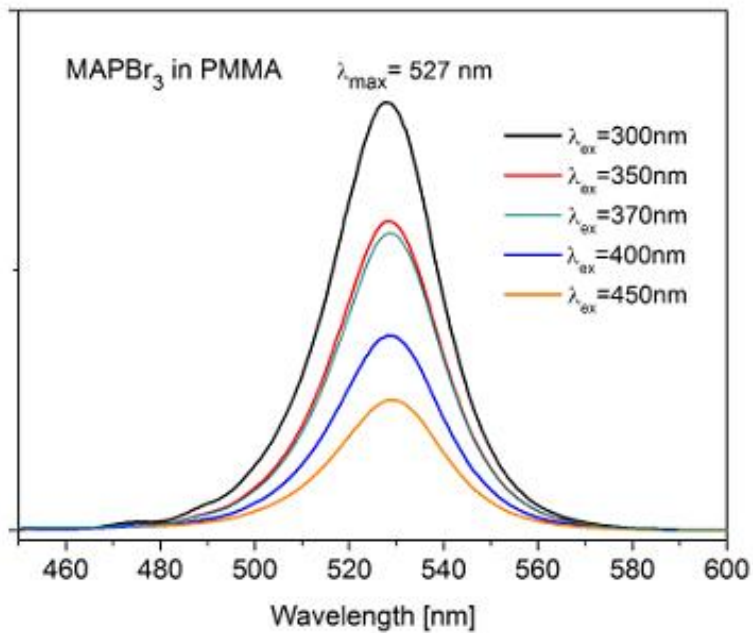


Nanoparticles  $\text{MAPbBr}_3$





# Light converter from perovskite nanoparticles





# Light converter from perovskite nanoparticles

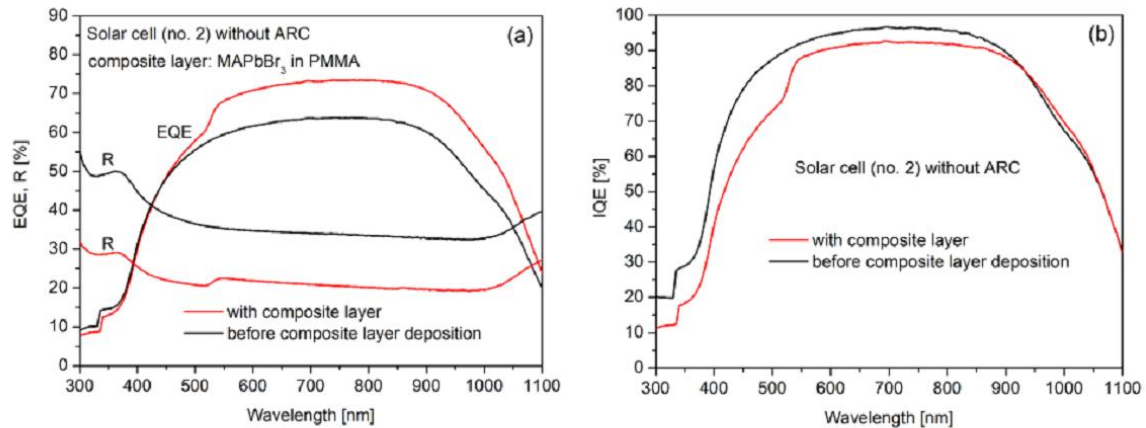
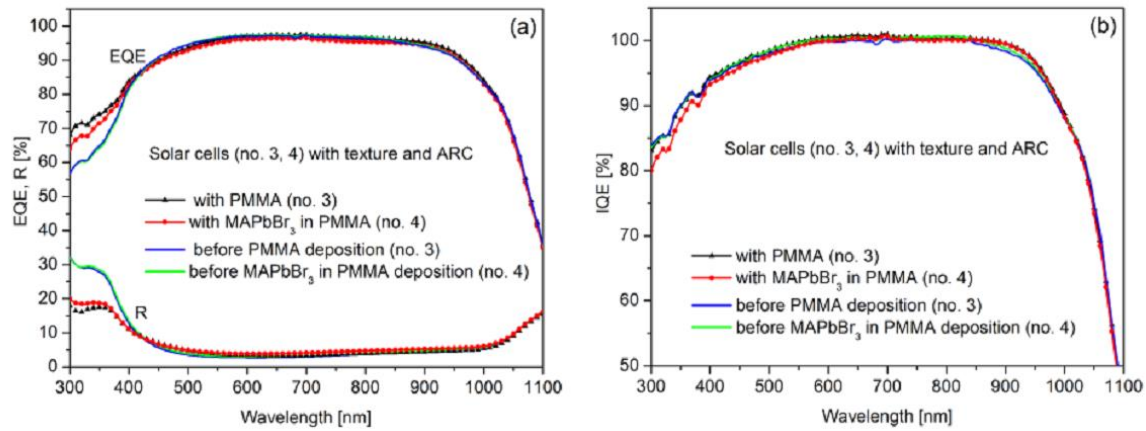


Fig. 8. The external EQE (a) and internal IQE (b) quantum efficiencies of the solar cell with and without the composite (MAPbBr<sub>3</sub> in PMMA) layer. The solar cell is without ARC. The difference between the IQE curves is caused by absorption of the composite layer

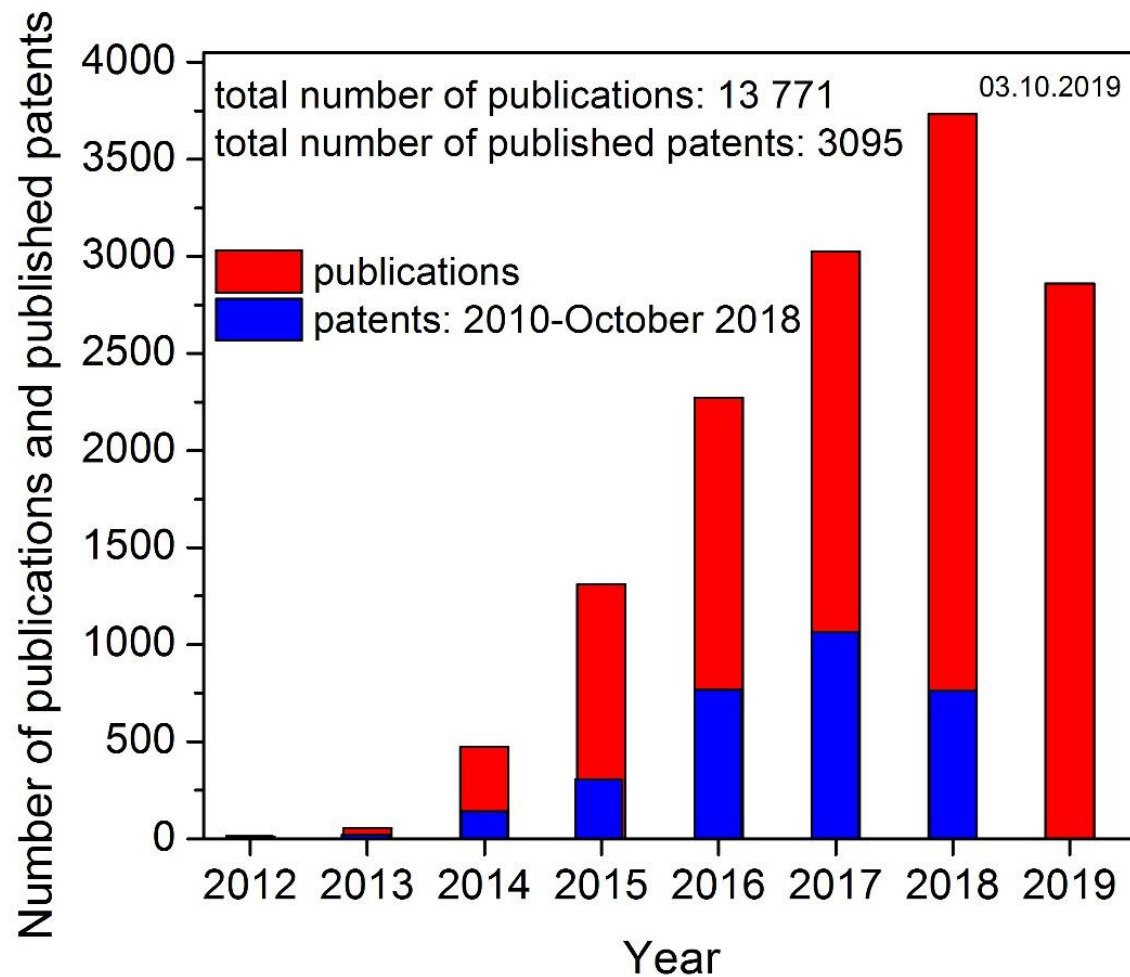




# Perovskite solar cells



# Progress in perovskite solar cells



Source: Web of Science. Topic: Perovskite solar cells

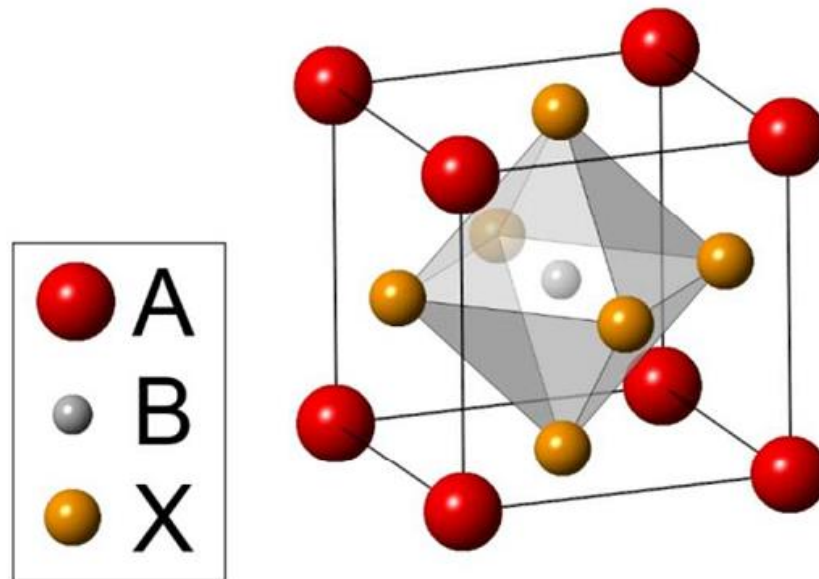
A review of the patent landscape. Cintelliq <https://go.nature.com/2IGsIR9> (2018).



# Perovskite $ABX_3$

**Perovskite - Calcium titanium oxide**  $CaTiO_3$  was discovered in the Urals Mountains in 1838 by Gustav Rose and named after Russian mineralogist Lev Perovski. All materials with the crystallographic structure of calcium titanium oxide  $CaTiO_3$  are named perovskites.

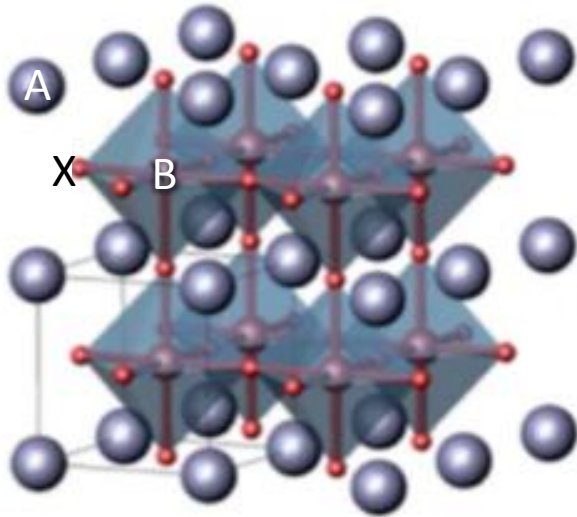
The general chemical formula for pure perovskite compounds is  $ABX_3$ , where 'A' and 'B' are two cations of very different sizes, and 'X' is an anion that binds to both.



Ideal crystal structure of cubic perovskite



# Halide perovskite $ABX_3$



## Perovskite crystalline system ( $ABX_3$ )

*Halide perovskite  $ABX_3$*   
 $A^I$  ( $Li^+$ ,  $K^+$ ,  $Cs^+$ ,  $CH_3NH_3^+$ )  
 $B^{II}$  ( $Pb^{2+}$ ,  $Mg^{2+}$ ,  $Ca^{2+}$ ,  $Sn^{2+}$ ,  $Ba^{2+}$ ,  $Zn^{2+}$ )  
 $X$  ( $F^-$ ,  $Cl^-$ ,  $Br^-$ ,  $I^-$ )

*Oxide perovskite  $ABO_3$*   
 $A^{II}$  ( $Mg^{2+}$ ,  $Ca^{2+}$ ,  $Ba^{2+}$ )  
 $B^{IV}$  ( $Ti^{4+}$ ,  $Si^{4+}$ )

*Organo-metal halide perovskite*

*Alkali-halide perovskite*

Peng Gao et al., Organohalide lead perovskites for photovoltaic applications, *Energy & Environmental Science*, 2014

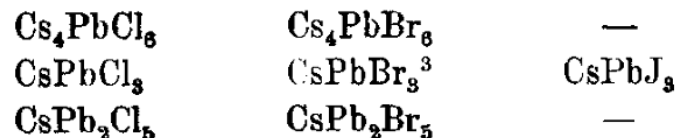


## Über die Cäsium- und Kalium-Bleihalogenide.

Von

H. L. WELLS.<sup>1</sup>

Als Fortsetzung der in diesem Laboratorium<sup>2</sup> begonnenen Arbeit über Doppelhalogenide ist von den Herren G. F. CAMPBELL, P. T. WALDEN und A. P. WHEELER eine Untersuchung über die Cäsium-Bleisalze unternommen worden. Diese Herren haben die Untersuchung mit vielem Eifer und Geschick durchgeführt, und es macht mir Freude, ihnen meinen Dank auszusprechen. Sie haben die Existenz folgender Salze konstatiert:



*Sheffield Scientific School, New Haven, Conn., Oktober 1892.*



# Halide perovskite $ABX_3$

X = F, Cl, Br, I

A = organic cation: MA ( $CH_3NH_3^+$ ), FA ( $CH_3(NH_2)_2^+$ ) or  $Cs^+$

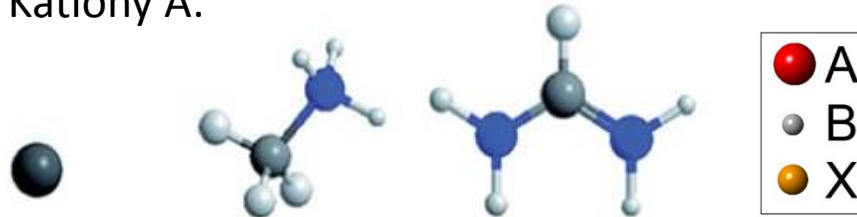
B = inorganic cations (Pb, Sn)

A	$R_A$ [nm]
MA	0,18
FA	0,19-0,22
Cs	0,17
Rb	0,15

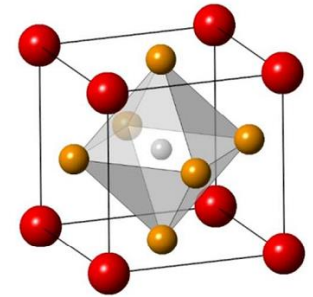
X	$R_A$ [nm]
I	0.220
Br	0.196
Cl	0.181

B	$R_B$ [nm]
Pb	0.119
Sn	0.110

Kationy A:



$Cs^+$  (cezu) MA (metyloamoniowy) FA (formamidinowy)



**Goldschmidts tolerance factor  $t$  :**

$$t = (R_B + R_X) / \{\sqrt{2}(R_B + R_X)\}$$

$R_A, R_B, R_X$  ionic radii

For halide perovskites:  
 $0,81 < t < 1,11$

$t = 0,89-1.0$  cubic structure  
 dla  $t < 0,89$  tetragonal or orthorhombic

M. A. Green, A. H-Bailie, and H. J. Snaith, Nature Photonics, **8**, 2014





# Halide perovskite $ABX_3$

	t	phase, color	Phase after annealing	$E_g$	PCE
$MAPbI_3$	0,89	Tetragonal, black	Tetragonal	1,5	20,3
$FAPbI_3$	1,02	Hexagonal, yellow	regular	1,49	17
$CsPbI_3$	0,79	Rhombic, yellow	Rhombic, yellow	1,72	10,77

$FAPbI_{3-x}Br_x$ ,  $E_g = 1.48 - 2.23$  eV



# Electronic properties of perovskites

Perovskit	Carriers	D(cm <sup>2</sup> s <sup>-1</sup> )	L <sub>D</sub> (nm)
CH <sub>3</sub> NH <sub>3</sub> Pbl <sub>3-x</sub> Cl <sub>x</sub>	electron	0.042±0.016	1069±204
	hole	0.054±0.022	1213±243
CH <sub>3</sub> NH <sub>3</sub> Pbl <sub>3</sub>	electron	0.017±0.011	129±41
	hole	0.011±0.007	105±32

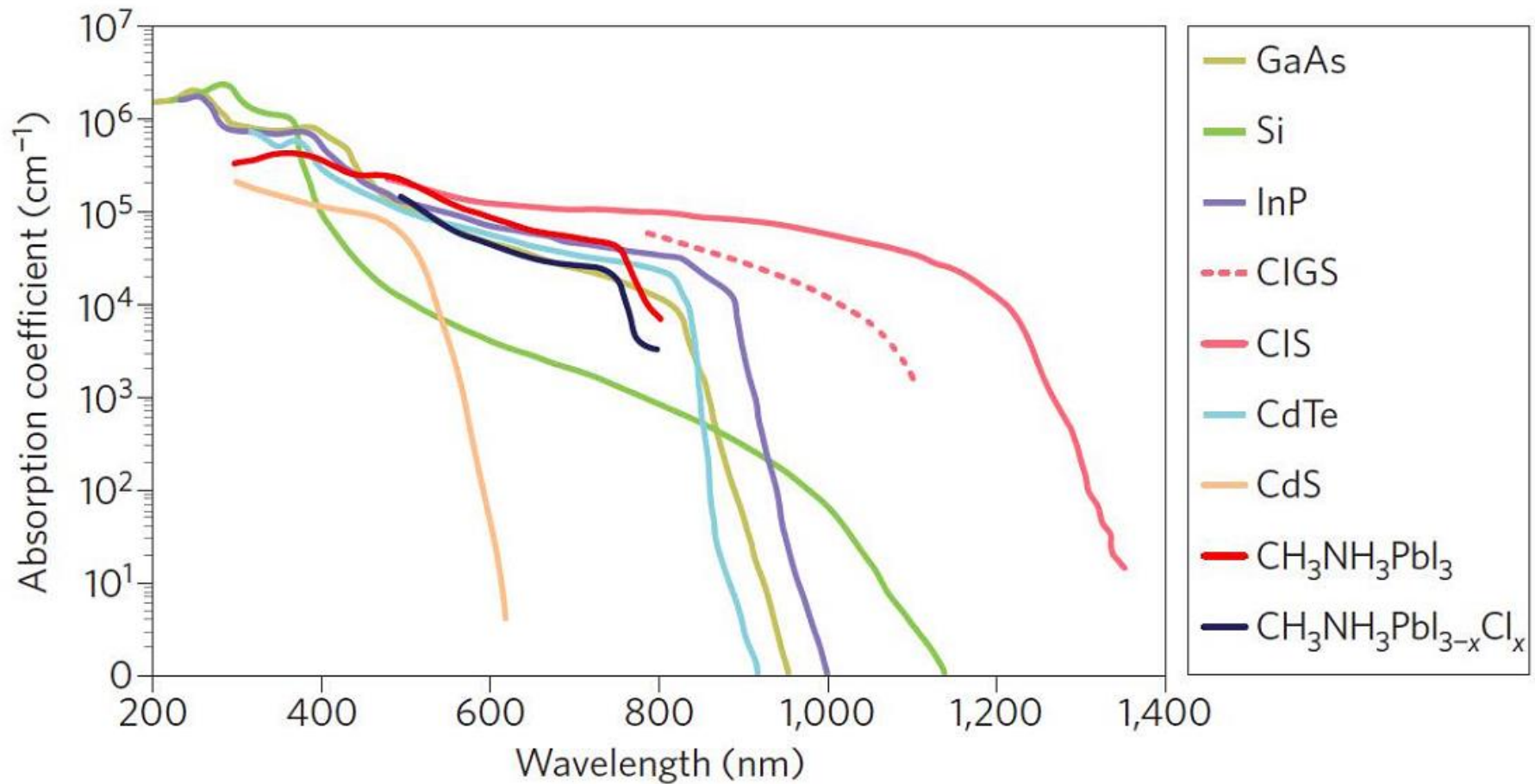
D diffusion coefficient

L<sub>D</sub> diffusion length

H.J. Snaith i współp.: *Science* 342 (2013) 341

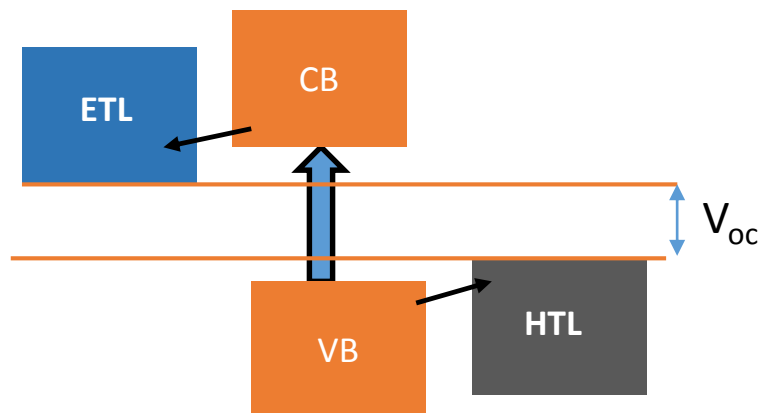


# Optical properties of perovskites





# Perovskite solar cells



ETL:  $\text{TiO}_2$ ,  $\text{SnO}_2$ , .....

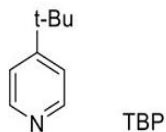
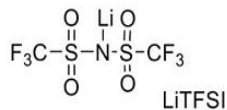
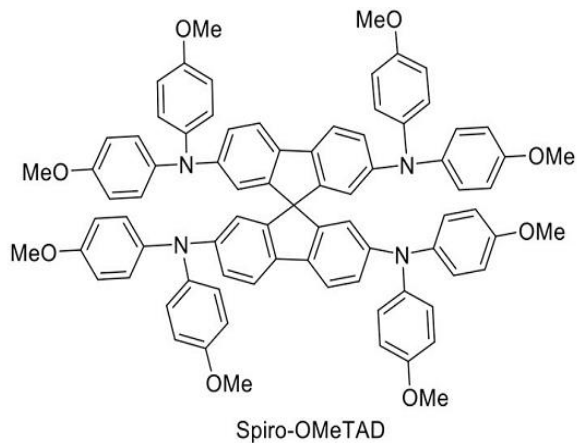
HTL: spiro-MeOTAD, PTAA

Absorber –  
Perowskit halogenkowy

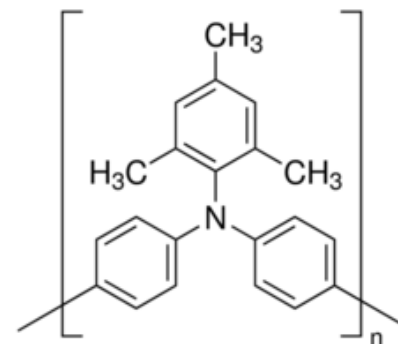
**spiro-MeOTAD =**

2,2',0,7,7'-tetrakis-(*N,N*-di-*p*-methoxyphenylamine)-9,9'-spirobifluorene)

LITFSI (lithium bis(trifluoromethanesulfonyl)imide) + TBP

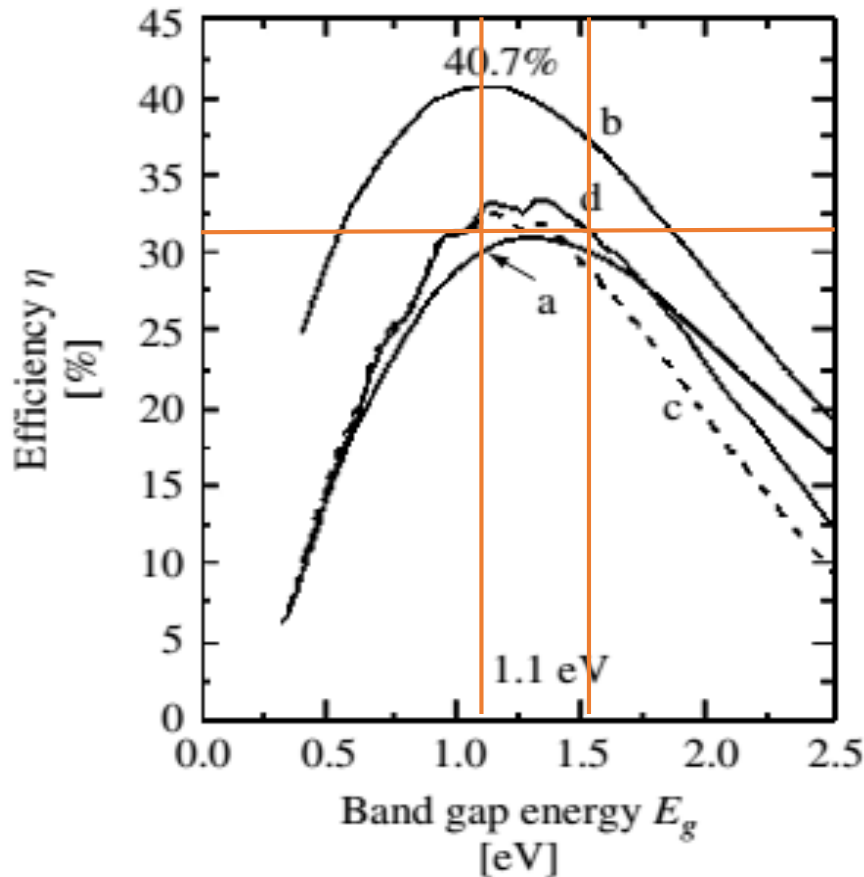


**PTAA - poly(triaryl amine)**





# Shockley - Queisser efficiency limit



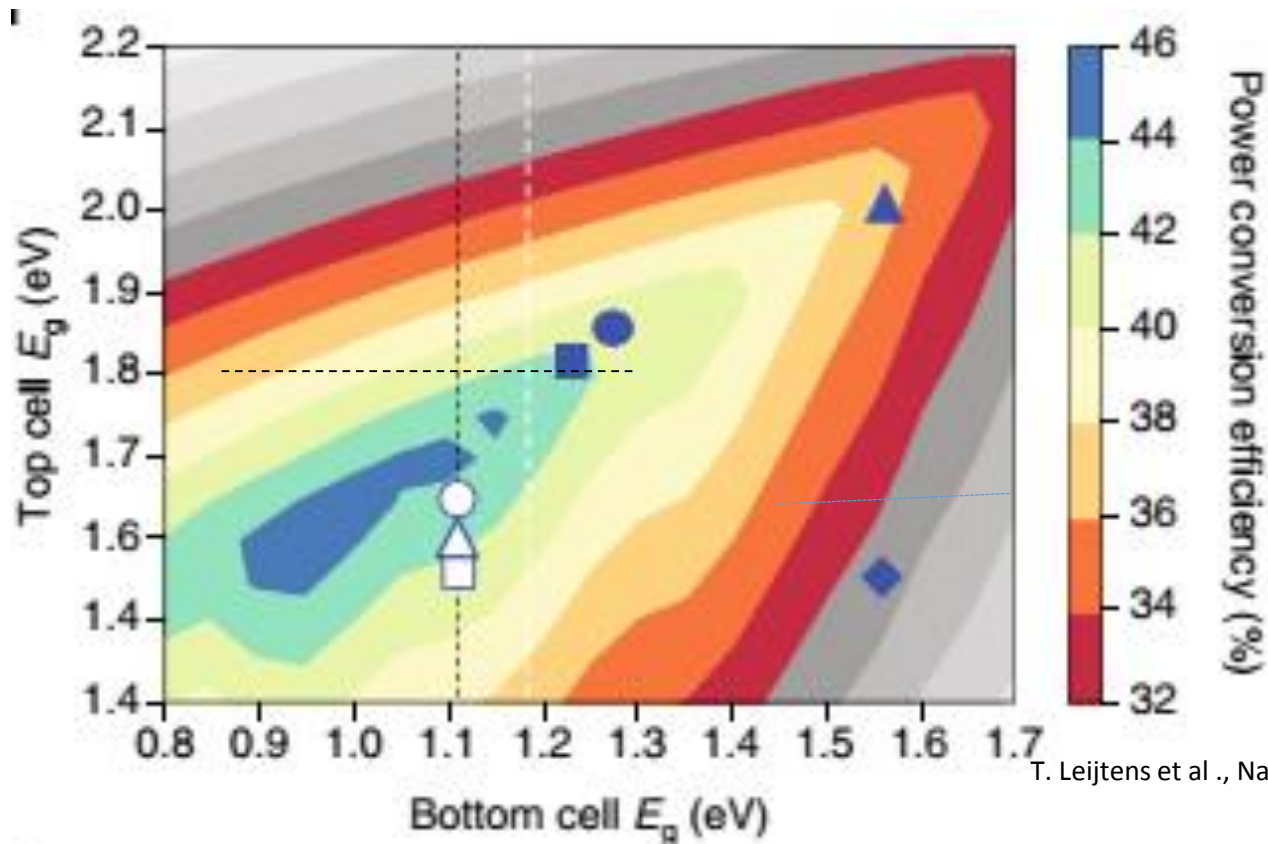
*Handbook of Photovoltaic Science and Engineering*, Ed: A. Luque and S. Hegedus, J. Wiley, 2003.

$E_g = 1.55$  eV for  $\text{MAPbI}_3$

Shockley - Queisser efficiency limit for an ideal solar cell versus band gap energy for: (a) unconcentrated 6000 K black body radiation ( $1595.9 \text{ Wm}^{-2}$ ); (b) full concentrated 6000 K black body radiation ( $7349.0 \times 10^4 \text{ Wm}^{-2}$ ); (c) unconcentrated AM1.5-Direct [18] ( $767.2 \text{ Wm}^{-2}$ ) and (d) **AM1.5 Global ( $962.5 \text{ Wm}^{-2}$ )**



# Theoretical limit for tandem (2-junctions)



T. Leijtens et al., Nature Energy, 3,(2018) 828–838

Top perovskite cell:  $\text{FA}_{0.83}\text{Cs}_{0.17}\text{Pb}(\text{I}_{0.6}\text{Br}_{0.4})_3$   $E_g = 1.72$  eV  
bottom cell Si:  $E_g = 1.12$  eV



## Advantages:

- Semiconductor with excellent opto-electronics properties,
- $E_g$  can be changed in wide range :1.2 - 2.0 eV,
- High absorption,
- Low non-radiative carrier recombination rates,
- Excellent charge transport: diffusion of length  $> 1\text{mm}$  )
- Low crystallization temperature
- Simple methods of manufacturing from solutions: spin-on, ink-jet printing, spray,
- Flexibility
- Earth-abundant elements: C, N, H, Pb, I..
- High efficiency  $> 20\%$

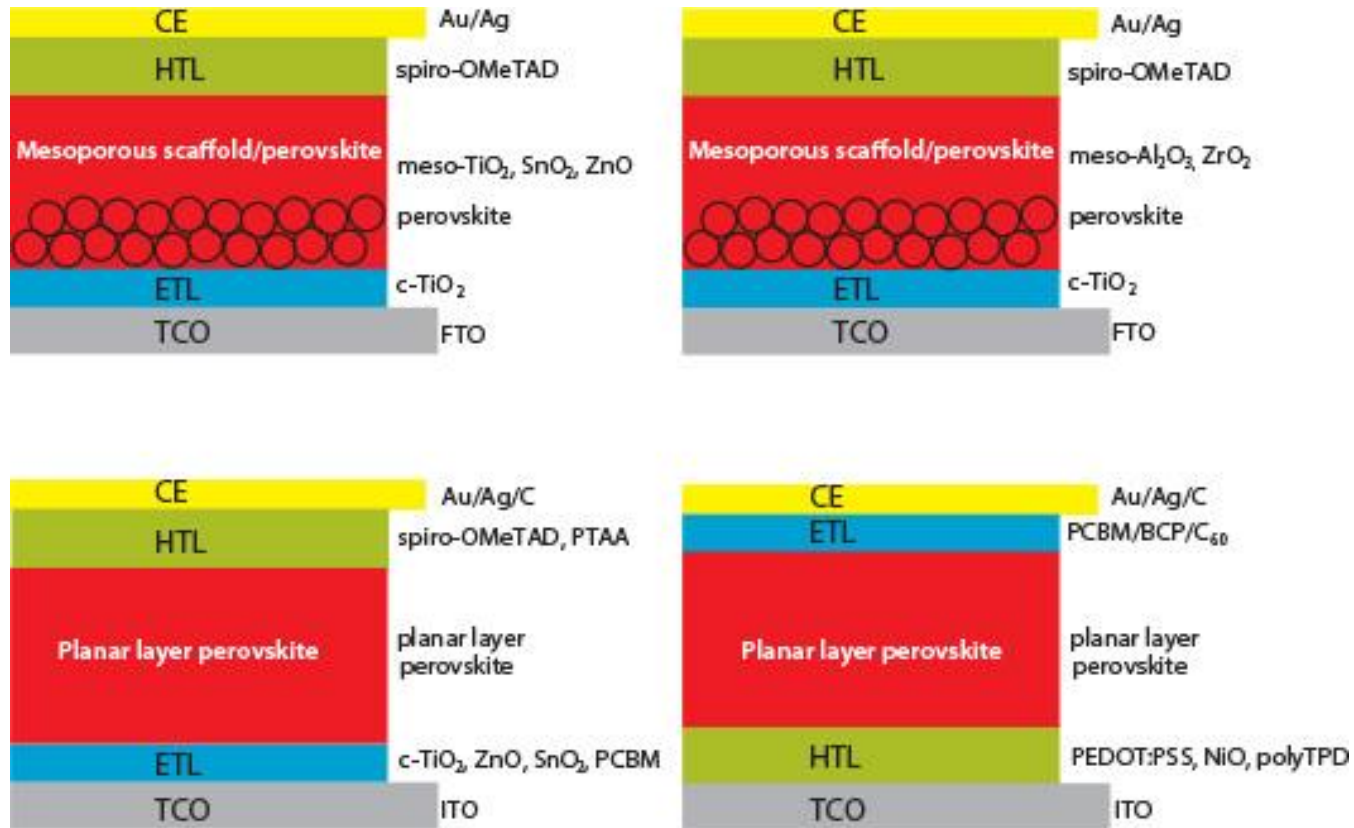
## Disadvanges:

- Low stability
- Toxicity from Pb

1. Eperon, G. E. et al. Perovskite-perovskite tandem photovoltaics with optimized bandgaps. *Science* 354, 861–865 (2016).
2. Eperon, G. E. et al. Formamidinium lead trihalide: A broadly tunable perovskite for efficient planar heterojunction solar cells. *Energy Environ. Sci.* 7, 982–988 (2014).
3. Unger, E. L. et al. Roadmap and road blocks for the band gap tunability of metal halide perovskites. *J. Mater. Chem. A* 5, 11401–11409 (2017).
5. H.J. Snaith et al., *Science* 342 (2013) 341



# Perovskite solar cells



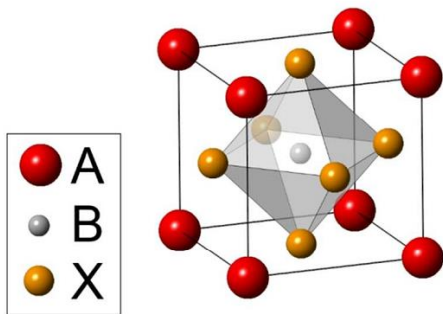
K. Kalyanasundaram, S. M. Zakeeruddin, M. Grätzel, *Material Matters*, 2016, 11.1, 3





# Halide perovskites $ABX_3$ and with mixed ions

$FAPbI_{3-x}Br_x$ ,  $E_g = 1.48 - 2.23$  eV



(B) Sn - decreases  $E_g$   
 (X) Br - increases  $E_g$

	t	$E_g$	PCE
MAPbI <sub>3</sub>	0.89	1.5	20.3
FAPbI <sub>3</sub>	1.02	1.49	17
CsPbI <sub>3</sub>	0.79	1.72	10,77
FA <sub>0.85</sub> MA <sub>0.15</sub> Pb(I <sub>0.85</sub> Br <sub>0.15</sub> ) <sub>3</sub>		1.62	22.1
FA <sub>0.85</sub> Cs <sub>0.15</sub> PbI <sub>3</sub>	0.99	1.52	17.3
FA <sub>0.85</sub> Cs <sub>0.17</sub> Pb(I <sub>0.83</sub> Br <sub>0.17</sub> ) <sub>3</sub>	1.01	1.74	20.0
FA <sub>0.75</sub> Cs <sub>0.25</sub> Pb <sub>0.5</sub> Sn <sub>0.5</sub> I <sub>3</sub>		1.2	

T. Leijtens et al. , *J. Mater. Chem. A*, 2017, 5,11483

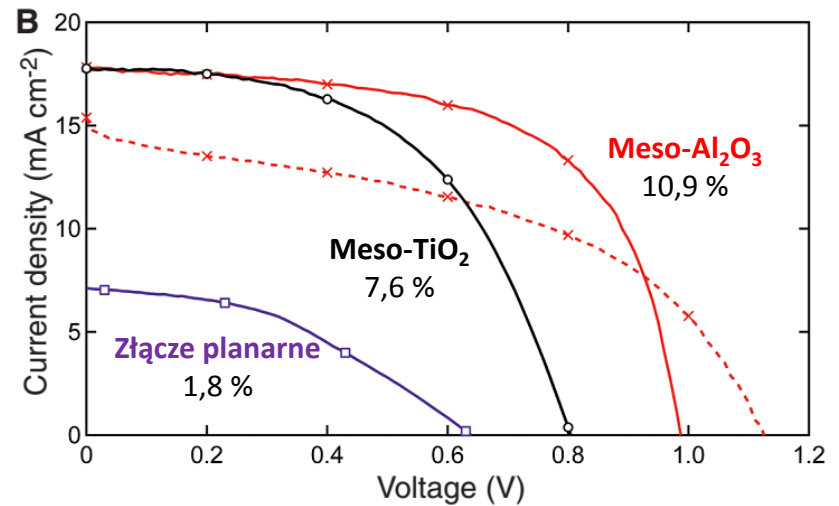
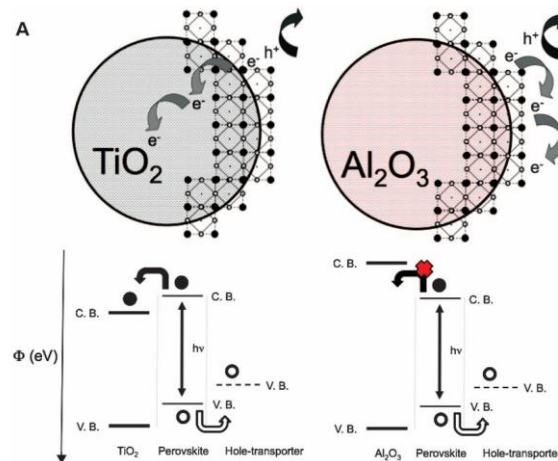


# Perovskite solar cells

## Efficient Hybrid Solar Cells Based on Meso-Superstructured Organometal Halide Perovskites

Michael M. Lee,<sup>1</sup> Joël Teuscher,<sup>1</sup> Tsutomu Miyasaka,<sup>2</sup> Takuro N. Murakami,<sup>2,3</sup> Henry J. Snaith<sup>1\*</sup>

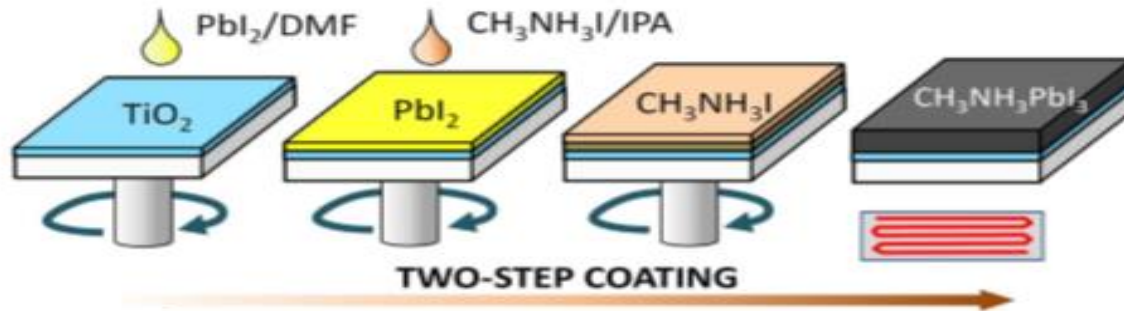
SCIENCE VOL 338 2 NOVEMBER 2012 643





# Perovskite solar cells

Two etap method



ng-Hyeok Im i in., Morphology-photovoltaic  
perty correlation in perovskite solar cells:  
e-step versus two-step deposition of  
 $\text{CH}_3\text{NH}_3\text{PbI}_3$ , APL Materials, 2014

**$\text{PbI}_2$**

Spin – coating: 3000 r. p. m, 30s

Annealing:  $40^\circ\text{C}$  – 2 min.,  $100^\circ\text{C}$  - 5 min.

**$\text{CH}_3\text{NH}_3\text{I}$**

Dippng: 40 s

Annealing :  $100^\circ\text{C}$ , 10 min

*Two-stage method - production of perovskites in mesoporous skeletal structures ( $\text{TiO}_x$ ,  $\text{ZnO}$ )*



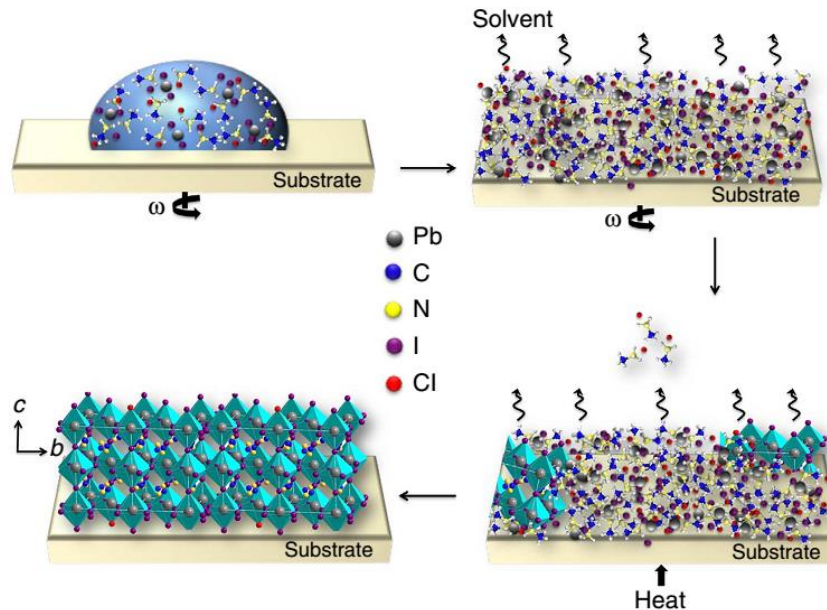
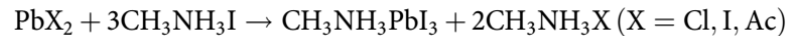
# Perovskite solar cells

one-step method

NATURE COMMUNICATIONS 2014

## Ultrasmooth organic-inorganic perovskite thin-film formation and crystallization for efficient planar heterojunction solar cells

Wei Zhang<sup>1</sup>, Michael Saliba<sup>1</sup>, David T. Moore<sup>2</sup>, Sandeep K. Pathak<sup>1</sup>, Maximilian T. Hörantner<sup>1</sup>, Thomas Stergiopoulos<sup>1</sup>, Samuel D. Stranks<sup>1</sup>, Giles E. Eperon<sup>1</sup>, Jack A. Alexander-Webber<sup>1</sup>, Antonio Abate<sup>1</sup>, Aditya Sadhanala<sup>3</sup>, Shuhua Yao<sup>4</sup>, Yulin Chen<sup>1</sup>, Richard H. Friend<sup>3</sup>, Lara A. Estroff<sup>2,5</sup>, Ulrich Wiesner<sup>2</sup> & Henry J. Snaith<sup>1</sup>





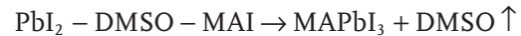
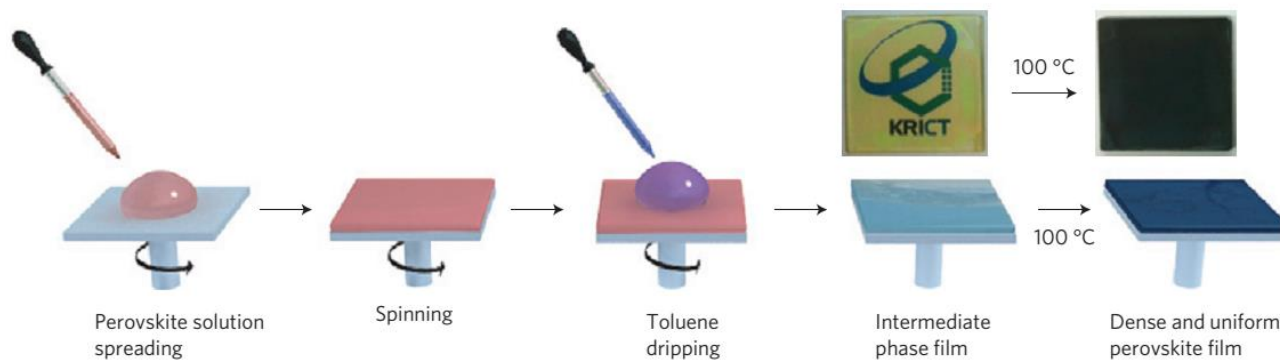
# Perovskite solar cells

**NATURE MATERIALS**, 2014

## Solvent engineering for high-performance inorganic-organic hybrid perovskite solar cells

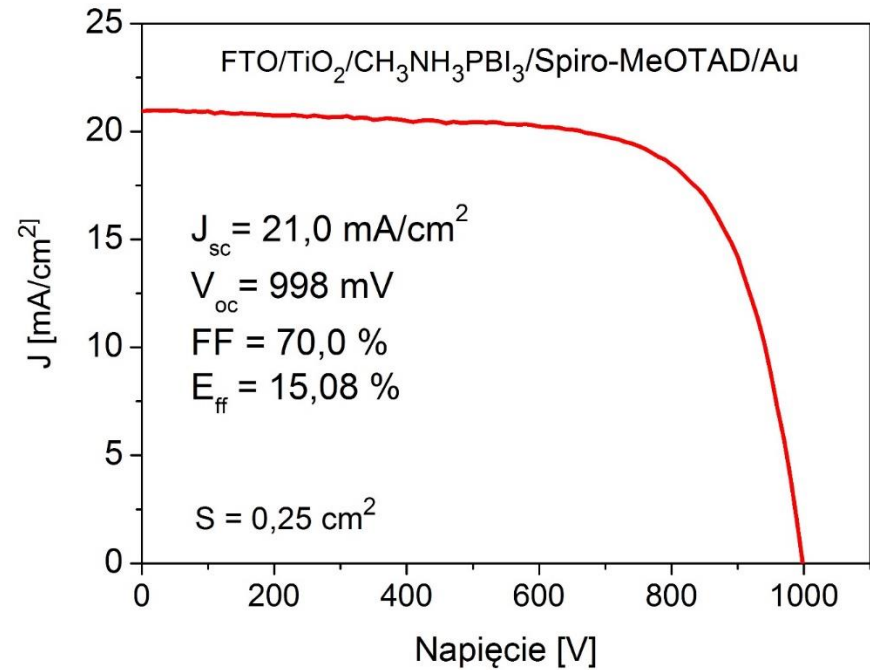
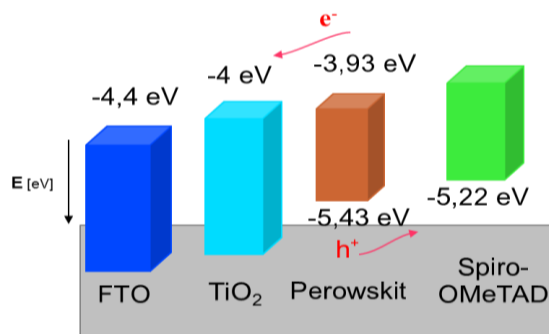
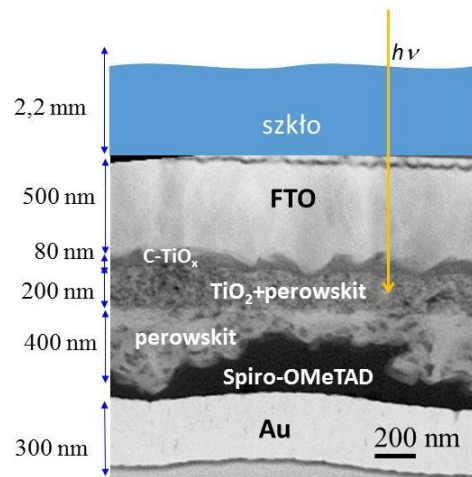
Nam Joong Jeon<sup>1†</sup>, Jun Hong Noh<sup>1†</sup>, Young Chan Kim<sup>1</sup>, Woon Seok Yang<sup>1</sup>, Seungchan Ryu<sup>1</sup> and Sang Il Seok<sup>1,2\*</sup>

Division of Advanced Materials, Korea Research Institute of Chemical Technology, Korea,  
Department of Energy Science, University, Suwon, Republic of Korea





# Perovskite solar cells

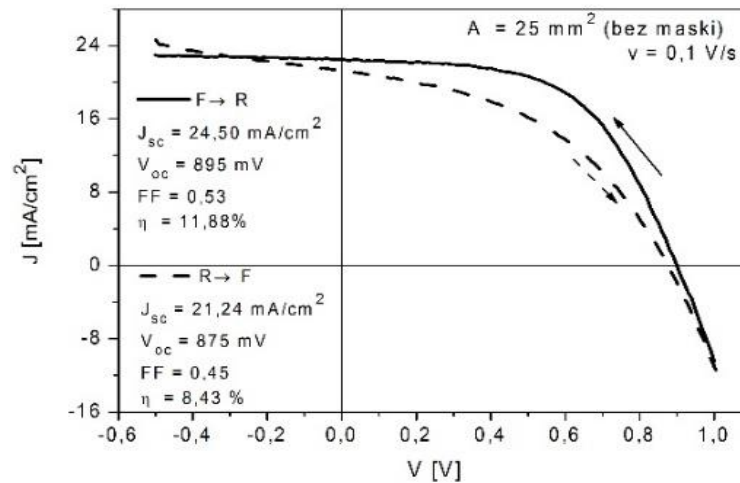
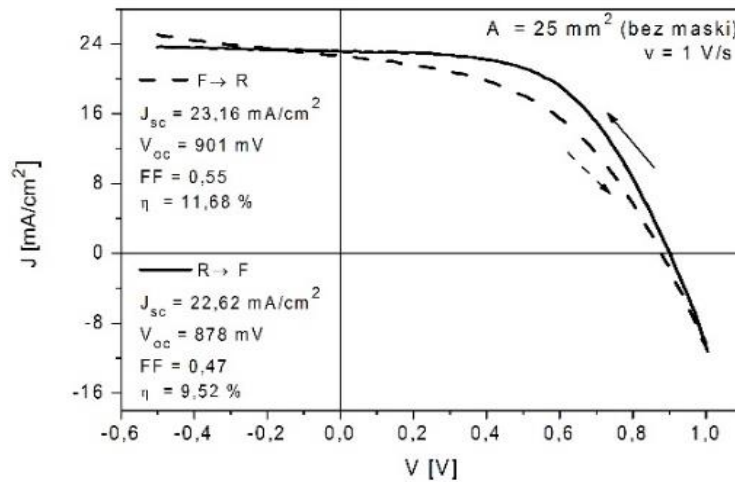
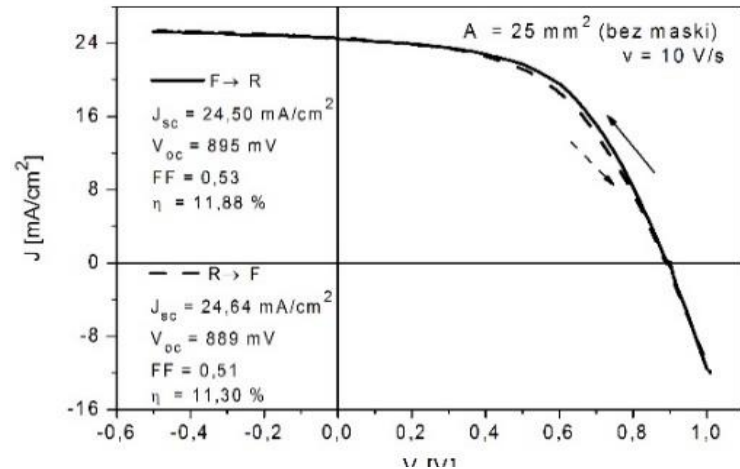
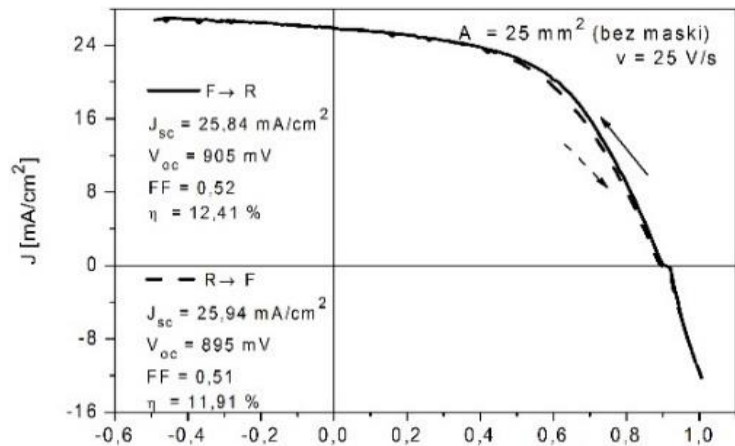


Characteristics of J-V perovskite cell made in LF IMIM PAN in Kozy

**Cells developed at LF IMIM**



# I-V characteristic hysteresis



J-V characteristics of the perovskite cell for both scan directions for scanning speeds of 25 V / s (a), 1V / s (b) and 0.5 V/s. Cells made in LF IMIM PAN in Kozy



# Stability

Opto-Electronics Review 25 (2017) 274–284



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Opto-Electronics Review

journal homepage: <http://www.journals.elsevier.com/opto-electronics-review>



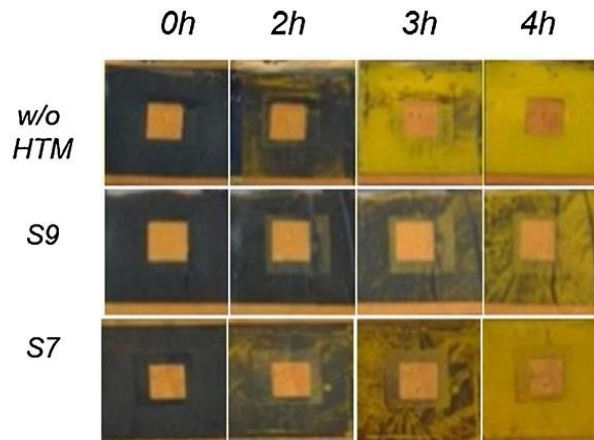
## Searching of new, cheap, air- and thermally stable hole transporting materials for perovskite solar cells

K. Gawlinska<sup>a,\*</sup>, A. Iwan<sup>b,\*</sup>, Z. Starowicz<sup>a</sup>, Grazyna Kulesza-Matlak<sup>a</sup>, K. Stan-Glowinska<sup>a</sup>, M. Janusz<sup>a</sup>, M. Lipinski<sup>a</sup>, B. Boharewicz<sup>c</sup>, I. Tazbir<sup>c</sup>, A. Sikora<sup>c</sup>

<sup>a</sup> Institute of Metallurgy and Materials Science, Polish Academy of Sciences, ul. Reymonta 25, 30-059 Krakow, Poland

<sup>b</sup> Military Institute of Engineer Technology, ul. Obornicka 136, 50-961 Wroclaw, Poland

<sup>c</sup> Electrotechnical Institute, Division of Electrotechnology and Materials Science, ul. M. Skłodowskiej-Curie 55/61, 50-369 Wroclaw, Poland



Destruction of  $\text{MAPbI}_3$  perovskite in the cell (without HTM and without encapsulation). A yellow color indicates the presence of  $\text{PbI}_2$ . The cell exposed to sunlight. Test of S9 and S7 polymers (polyazomethines) for encapsulation



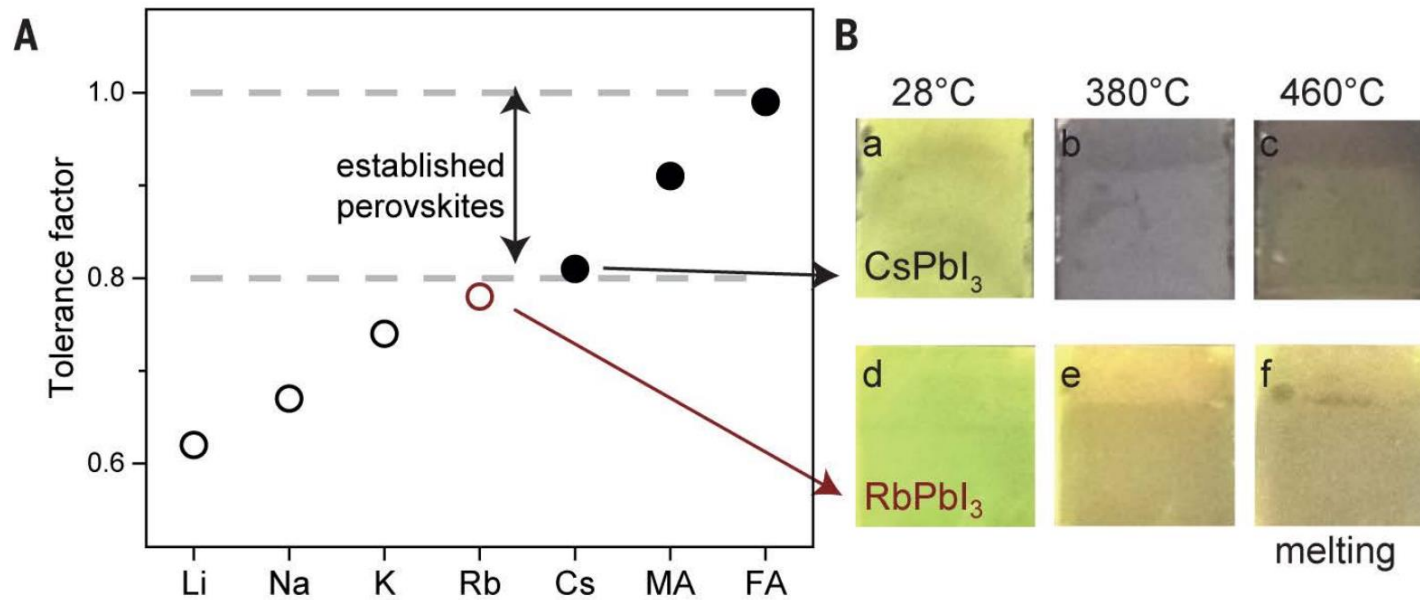


# Stability

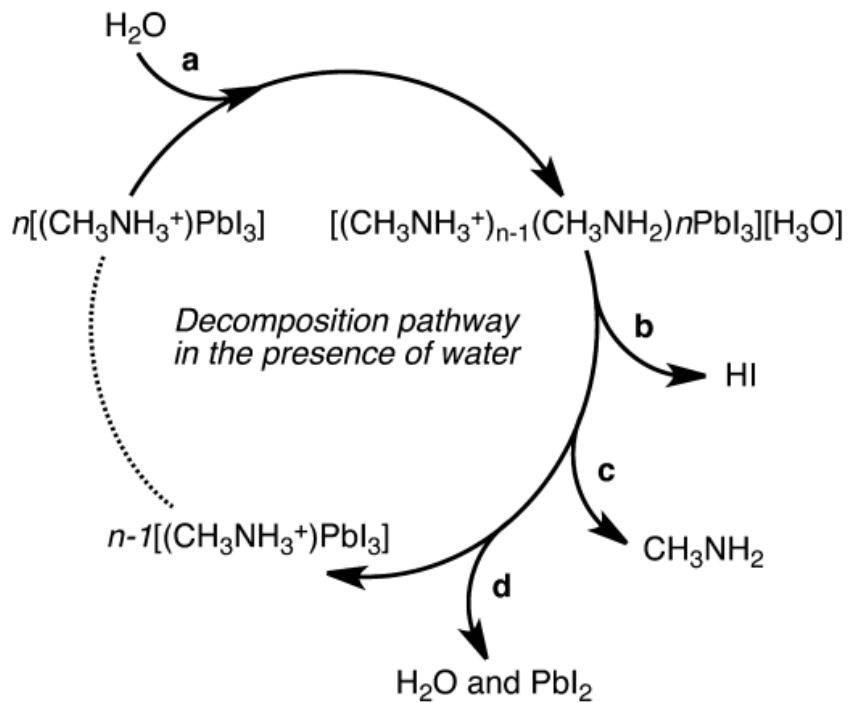
- Structure stability
- Thermal stability
- Atmospheric stability
- Oxygen interaction
- Water impact
- Stability to UV radiation



# Stability



M. Saliba, T. Matsui, K. Domanski, J.-Y. Seo, A. Ummadisingu, S. M. Zakeeruddin, J.-P. Correa-Baena, W. R. Tress, A. Abate and A. Hagfeldt, Incorporation of Rubidium Cations into Perovskite Solar Cells Improves Photovoltaic Performance, *Science*, 2016, 354(6309), 206.



Another mechanism according to [2]

CH<sub>3</sub>NH<sub>2</sub> methylamine - volatile and water-soluble HI-water-soluble 1]

1. J. M. Frost, K. T. Butler, F. Brivio, C. H. Hendon, M. Van Schilfgaarde and A. Walsh,, Nano Lett., 2014, 14, 2584–2590
2. J. A. Christians , P. A. Miranda Herrera , P. V. Kamat ,J. Am. Chem. Soc. 2015 ,137 , 1530 .



# Thermal stability

## Requirements:

Cell operating temperature -40 to > 85 °C. Cell operation up to 85 °C

Lamination - 150 °C

**MAPbX<sub>3</sub> unstable at 85°C - MA sublimates at 85 °C even in an inert atmosphere**

Materials  
Views

[www.MaterialsViews.com](http://www.MaterialsViews.com)

Adv. Energy Mater., 2015, 1500477

ADVANCED  
ENERGY  
MATERIALS

[www.advenergymat.de](http://www.advenergymat.de)

## Intrinsic Thermal Instability of Methylammonium Lead Trihalide Perovskite

*Bert Conings,\* Jeroen Drijkoningen, Nicolas Gauquelin, Aslihan Babayigit, Jan D'Haen, Lien D'Olieslaeger, Anitha Ethirajan, Jo Verbeeck, Jean Manca, Edoardo Mosconi, Filippo De Angelis, and Hans-Gerd Boyen\**

Institute for Materials Research  
Hasselt University  
Wetenschapspark 1, 3590 Diepenbeek, Belgium

Electron Microscopy for Materials Research (EMAT)  
University of Antwerp  
Groenenborgerlaan 171, 2020 Antwerp, Belgium

X-LaB, Hasselt University  
Agoralaan Building D, 3590 Diepenbeek, Belgium

Computational Laboratory for Hybrid/Organic Photovoltaics (CLHYO)  
CNR-ISTM, I-06123 Perugia, Italy

**MAPbX<sub>3</sub> is not suitable for industrial production!**

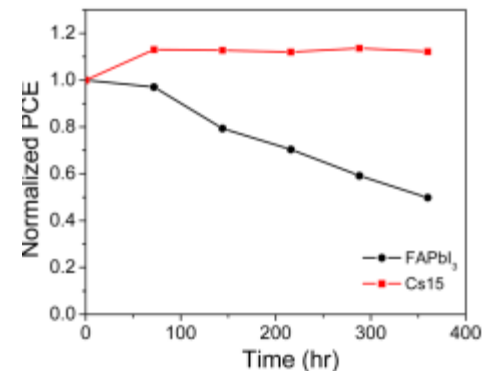
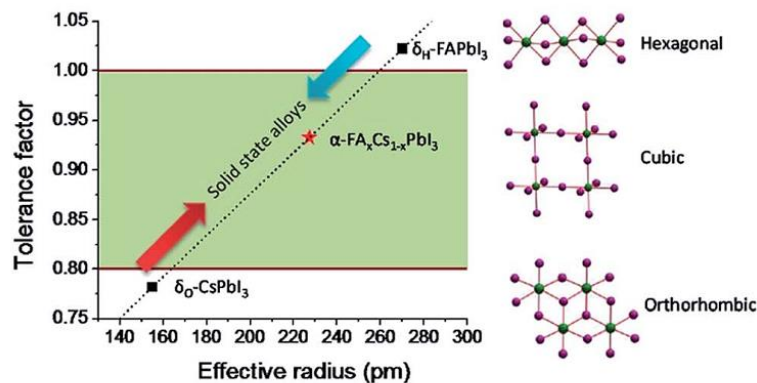


# Stabilizing Perovskite Structures by Tuning Tolerance Factor: Formation of Formamidinium and Cesium Lead Iodide Solid-State Alloys

Zhen Li,<sup>†</sup> Mengjin Yang,<sup>†</sup> Ji-Sang Park,<sup>†</sup> Su-Huai Wei,<sup>†,§</sup> Joseph J. Berry,<sup>†</sup> and Kai Zhu<sup>\*,†</sup>

<sup>†</sup>National Renewable Energy Laboratory, Golden, Colorado 80401, United States

<sup>§</sup>Beijing Computational Science Research Center, Beijing 100094, China





## 3D metal halide perovskites used in photovoltaics

### Mixed cations [FAMA], [FACs]

	t	Faza, kolor	Faza po wygrzaniu	Eg	PCE	Przejście fazowe
MAPbI <sub>3</sub>	0,89	Tetragonal, black	Tetragonal	1,5	20,3	Regular, 60° C
FAPbI <sub>3</sub>	1,02	Hexagonal, yellow	regular	1,49	17	Regular, 150° C
CsPbI <sub>3</sub>	0,79	Rhombic, yellow	Rhombic, yellow	1,72	10,77	Regular, 300° C
FA <sub>0.85</sub> MA <sub>0.15</sub> Pb(I <sub>0.85</sub> Br <sub>0.15</sub> ) <sub>3</sub>		Regular, black	Regular black	1,62	22,1	
FA <sub>0.85</sub> Cs <sub>0.15</sub> PbI <sub>3</sub>	0,99	Tetragonal, black	tetragonal	1,52	17,3	
FA <sub>0.85</sub> Cs <sub>0.15</sub> Pb(I <sub>0.83</sub> Br <sub>0.17</sub> ) <sub>3</sub>	1,01	Tetragonal, blacka	tetragonal	1,74	20,0	

Tomas Leijtens, Kevin Bush, Rongrong Cheacharoen, Rachel Beal,

Andrea Bowring and Michael D. McGehee,

*Towards enabling stable lead halide perovskite solar cells; interplay between structural, environmental, and thermal stability*, J. Mater. Chem. A, 2017, 5,11483

Department of Materials Science, Stanford University, Lomita Mall, Stanford, CA, USA.



# 3D metal halide perovskites used in photovoltaics

## Mixed cations [FAMA], [FACs]

	Eg [eV]	PCE [%]	cell	ref
$\text{FA}_{0.83}\text{Cs}_{0.17}\text{Pb}(\text{I}_{0.83}\text{Br}_{0.17})_3$	1,74	23,6 PSC/Si	Tandem monolit PSC/Si	[2]
$\text{FA}_{0.75}\text{Cs}_{0.25}\text{Pb}_{0.5}\text{Sn}_{0.5}\text{I}_3$	1,2	17,0 PSC/PSC	Tandem PSC/PSC	[3]
$\text{FA}_{0.83}\text{Cs}_{0.17}\text{Pb}(\text{I}_{0.5}\text{Br}_{0.5})_3$	1,8			

### [1] High-performance photovoltaic perovskite layers fabricated through intramolecular exchange

Woon Seok Yang,<sup>1\*</sup> Jun Hong Noh,<sup>1\*</sup> Nam Joong Jeon,<sup>1</sup> Young Chan Kim,<sup>1</sup>  
Seungchan Ryu,<sup>1</sup> Jangwon Seo,<sup>1</sup> Sang Il Seok<sup>1,2†</sup>

<sup>1</sup>Division of Advanced Materials, Korea Research Institute of Chemical Technology, 141 Gajeong-Ro, Yuseong-Gu, Daejeon 305-600, Korea. <sup>2</sup>Department of Energy Science, Sungkyunkwan University, Suwon 440-746, Korea.

*Science*, 2015, 348(6240),1234

### [2] NREL, *Best Research-Cell Efficiencies*, 2017



[2]

## 23.6%-efficient monolithic perovskite/silicon tandem solar cells with improved stability

Kevin A. Bush<sup>1†</sup>, Axel F. Palmstrom<sup>1†</sup>, Zhengshan J. Yu<sup>2†</sup>, Mathieu Boccard<sup>2</sup>, Rongrong Cheacharoen<sup>1</sup>, Jonathan P. Mailoa<sup>3</sup>, David P. McMeekin<sup>4</sup>, Robert L. Z. Hoye<sup>3</sup>, Colin D. Bailie<sup>1</sup>, Tomas Leijtens<sup>1</sup>, Ian Marius Peters<sup>3</sup>, Maxmillian C. Minichetti<sup>1</sup>, Nicholas Rolston<sup>1</sup>, Rohit Prasanna<sup>1</sup>, Sarah Sofia<sup>3</sup>, Duncan Harwood<sup>5</sup>, Wen Ma<sup>6</sup>, Farhad Moghadam<sup>6</sup>, Henry J. Snaith<sup>4</sup>, Tonio Buonassisi<sup>3</sup>, Zachary C. Holman<sup>2\*</sup>, Stacey F. Bent<sup>1</sup> and Michael D. McGehee<sup>1\*</sup>

<sup>1</sup>Stanford University, Stanford 94305, USA. <sup>2</sup>Arizona State University, Tempe 85281, USA. <sup>3</sup>Massachusetts Institute of Technology, Cambridge 02139, USA. <sup>4</sup>University of Oxford, Oxford OX1 3PU, UK. <sup>5</sup>D2 Solar LLC, San Jose 95131, USA. <sup>6</sup>Sunpreme, Sunnyvale 94085, USA. <sup>†</sup>These authors contributed equally to this work. \*e-mail: [Zachary.holman@asu.edu](mailto:Zachary.holman@asu.edu); [Mmcgehee@stanford.edu](mailto:Mmcgehee@stanford.edu)

[3]

Science

REPORTS

Cite as: G. E. Eperon *et al.*, *Science*  
10.1126/science.aaf9717 (2016).

## Perovskite-perovskite tandem photovoltaics with optimized bandgaps

Giles E. Eperon,<sup>1,3\*</sup> Tomas Leijtens,<sup>2\*</sup> Kevin A. Bush,<sup>2</sup> Rohit Prasanna,<sup>2</sup> Thomas Green,<sup>1</sup> Jacob Tse-Wei Wang,<sup>1</sup> David P. McMeekin,<sup>1</sup> George Volonakis,<sup>4</sup> Rebecca L. Milot,<sup>1</sup> Richard May,<sup>2</sup> Axel Palmstrom,<sup>2</sup> Daniel J. Slotcavage,<sup>2</sup> Rebecca A. Belisle,<sup>2</sup> Jay B. Patel,<sup>1</sup> Elizabeth S. Parrott,<sup>1</sup> Rebecca J. Sutton,<sup>1</sup> Wen Ma,<sup>5</sup> Farhad Moghadam,<sup>5</sup> Bert Conings,<sup>1,6</sup> Aslihan Babayigit,<sup>1,6</sup> Hans-Gerd Boyen,<sup>6</sup> Stacey Bent,<sup>2</sup> Feliciano Giustino,<sup>4</sup> Laura M. Herz,<sup>1</sup> Michael B. Johnston,<sup>1</sup> Michael D. McGehee,<sup>2†</sup> Henry J. Snaith<sup>1†</sup>

<sup>1</sup>Department of Physics, University of Oxford, Clarendon Laboratory, Parks Road, Oxford OX1 3PU, UK. <sup>2</sup>Department of Materials Science, Stanford University, Lomita Mall, Stanford, CA, USA. <sup>3</sup>Department of Chemistry, University of Washington, Seattle, WA, USA. <sup>4</sup>Department of Materials, University of Oxford, Parks Road, Oxford OX1 3PU, UK. <sup>5</sup>SunPreme, Palomar Avenue, Sunnyvale, CA, USA. <sup>6</sup>Institute for Materials Research, Hasselt University, Diepenbeek, Belgium.





Cite as: M. Saliba *et al.*, *Science*  
10.1126/science.aah5557 (2016).

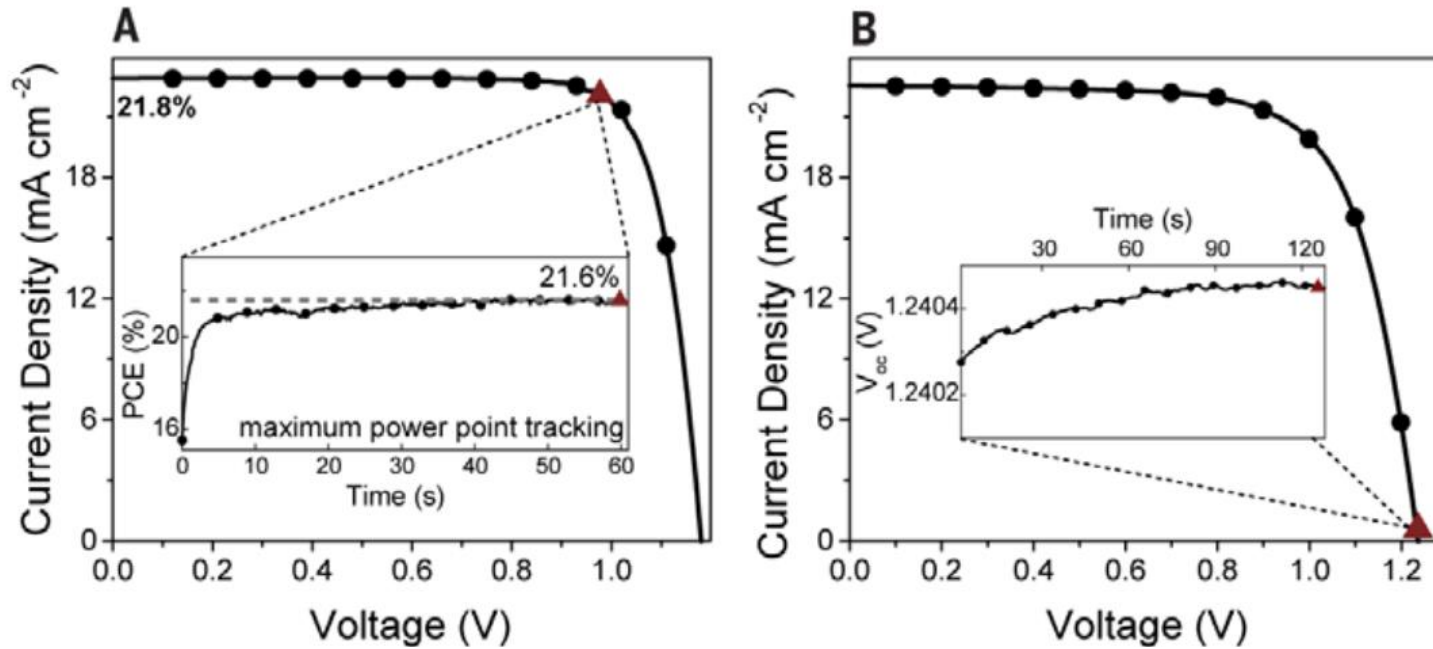
## Incorporation of rubidium cations into perovskite solar cells improves photovoltaic performance

**Michael Saliba,<sup>1\*†</sup> Taisuke Matsui,<sup>1,2\*</sup> Konrad Domanski,<sup>1\*</sup> Ji-Youn Seo,<sup>1</sup> Amita Ummadisingu,<sup>1</sup> Shaik M. Zakeeruddin,<sup>1</sup> Juan-Pablo Correa-Baena,<sup>3</sup> Wolfgang R. Tress,<sup>1</sup> Antonio Abate,<sup>1</sup> Anders Hagfeldt,<sup>3</sup> Michael Grätzel<sup>1†</sup>**

<sup>1</sup>Laboratory of Photonics and Interfaces, École Polytechnique Fédérale de Lausanne, Station 6, CH-1015 Lausanne, Switzerland. <sup>2</sup>Advanced Research Division, Materials Research Laboratory, Panasonic Corporation, 1006 Kadoma, Kadoma City, Osaka 571-8501, Japan. <sup>3</sup>Laboratory of Photomolecular Science, École Polytechnique Fédérale de Lausanne, Station 6, CH-1015 Lausanne, Switzerland.



# Mixed cations [RbCsMAFA]

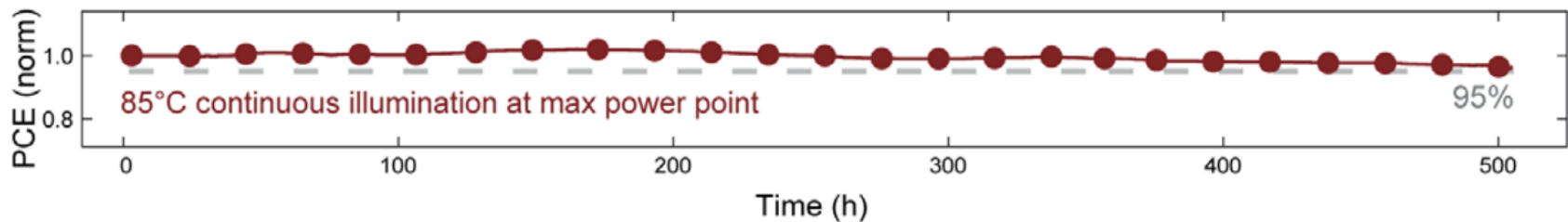


- A) J-V characteristic for 10 mVs<sup>-1</sup> cell with 21.8% efficiency (V<sub>oc</sub> = 1180 mV, J<sub>sc</sub> = 22.8 mA cm<sup>-2</sup>, FF 81%).
- B) cell with the highest V<sub>oc</sub>. 19% PCE stabilized for 0.5 cm<sup>2</sup> cell.



# Mixed cations [RbCsMAFA]

## Kationy mieszane [RbCsMAFA]



Thermal stability test. Aging 500 hours at 85°C, full solar lighting at the point of maximum power in the atmosphere N<sub>2</sub>. Aging procedure more stringent than for industrial standards.



# Layered perovskites

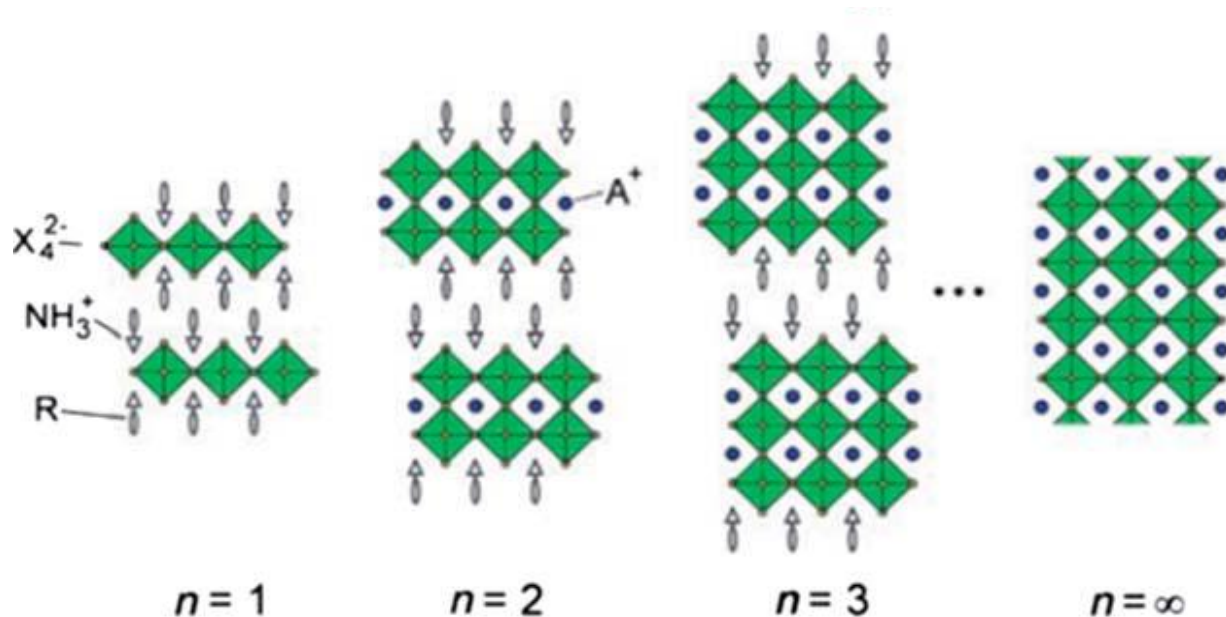
$R_2(A)_{n-1}B_nX_{3n+1}$  ( $n=1, 2, 3, 4, \dots$ )  $n$  the number of layer (Ruddlesden-Popper structure).

Dla  $R$ = kation butyloamoniowy ( $n$  butylammonium)

$R$  – large alkylammonium cations:

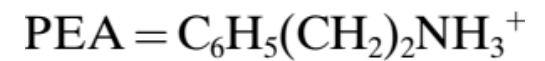
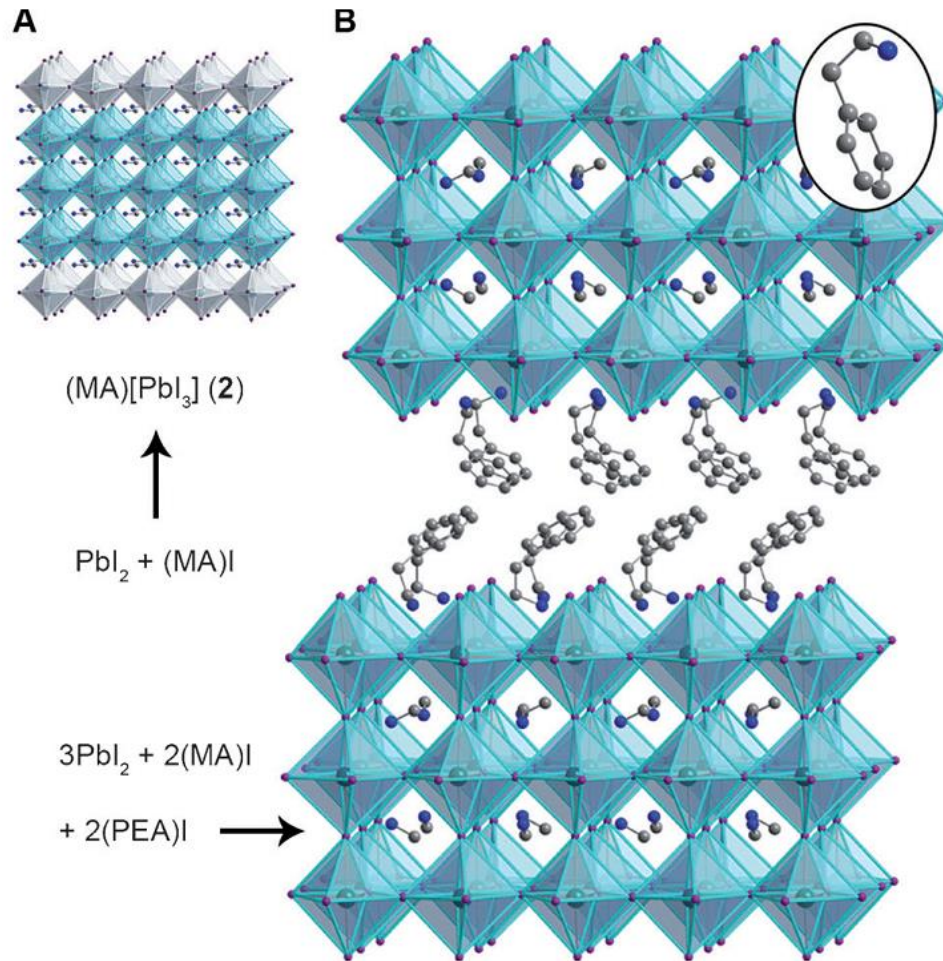
PEA =  $C_8H_9NH_3$  phenylethylammonium cation

BA =  $C_4H_9NH_3$  butylammonium cation





# Layered perovskites



$$V_{\text{oc}} = 1.18 \text{ V}$$
$$\text{PCE} = 4.73\%$$

$$E_{\text{g}} = 2.1 \text{ eV}$$



## Efficient ambient-air-stable solar cells with 2D-3D heterostructured butylammonium-caesium-formamidinium lead halide perovskites

Zhiping Wang, Qianqian Lin, Francis P. Chmiel, Nobuya Sakai, Laura M. Herz and Henry J. Snaith\*

Clarendon Laboratory, Department of Physics, University of Oxford, Parks Road, Oxford OX1 3PU, UK.

3D perovskite  $\text{FA}_{0.83}\text{Cs}_{0.17}\text{Pb}(\text{I}_y\text{Br}_{1-y})_3$

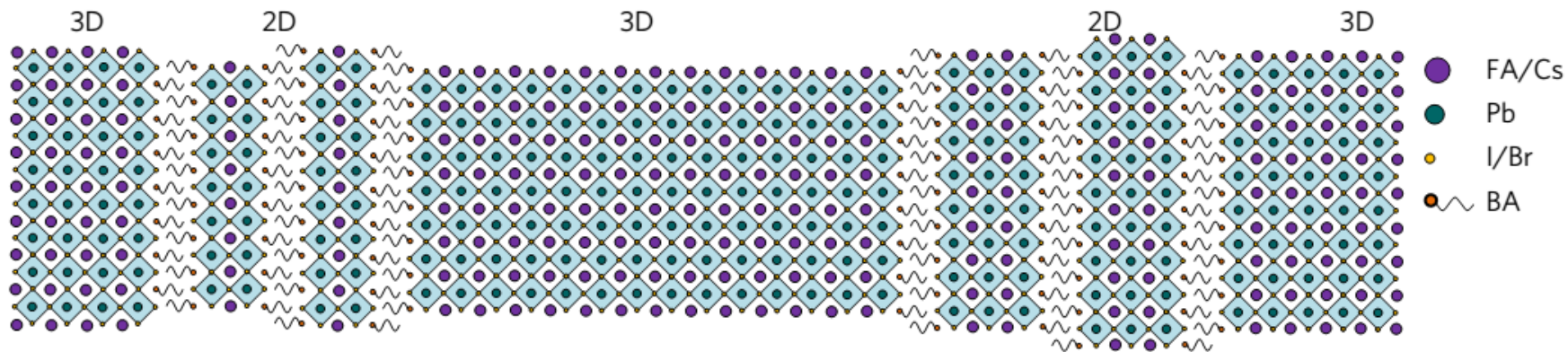
2D perovskite  $(\text{BA})_2(\text{MA})_3\text{Pb}_4\text{I}_{13}$

2D-3D structure:

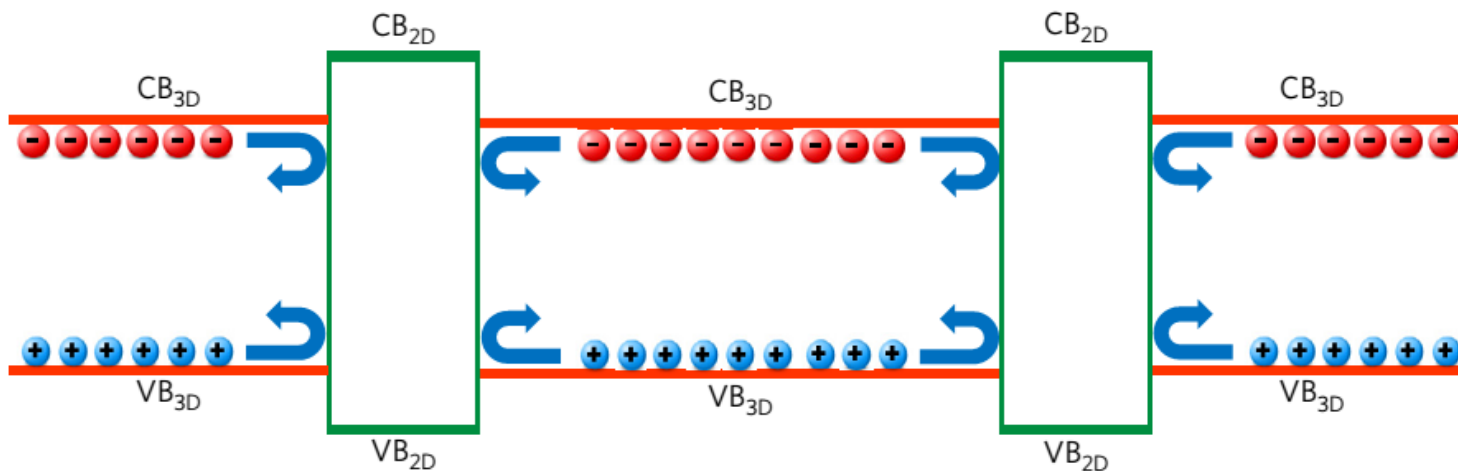
$\text{BA}_{0.09}(\text{FA}_{0.83}\text{Cs}_{0.17})_{0.91}\text{Pb}(\text{I}_{0.6}\text{Br}_{0.4})_3$  ( $x = 0.09$ )



# 2D-3D structures



Model struktury 2D-3D

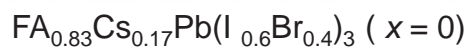
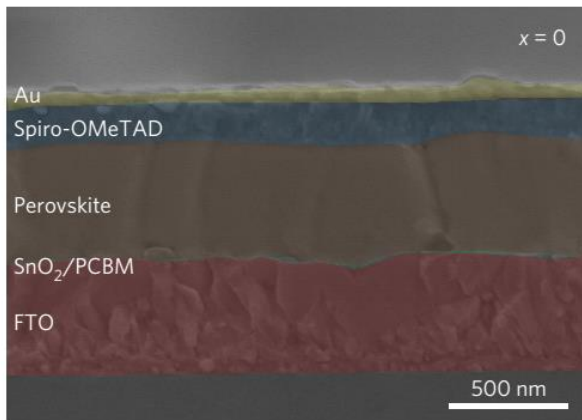


Model of energetic bands of 2D-3D structure, CB- conductivity band, VB valence band.

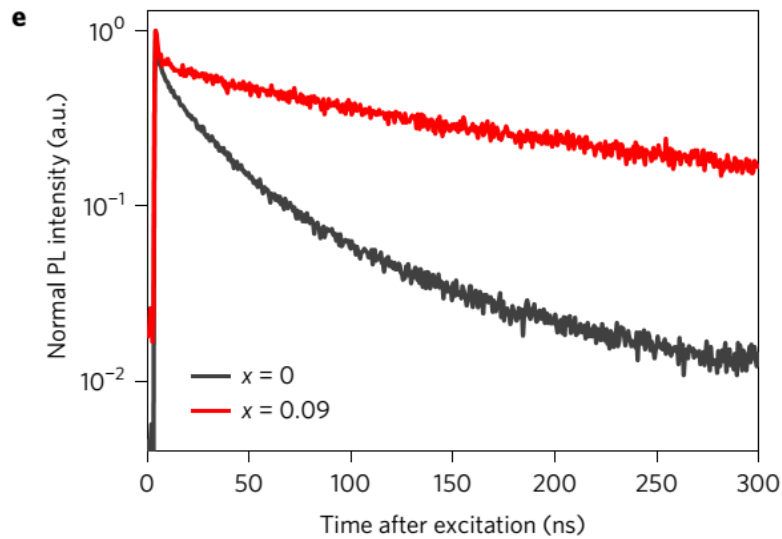
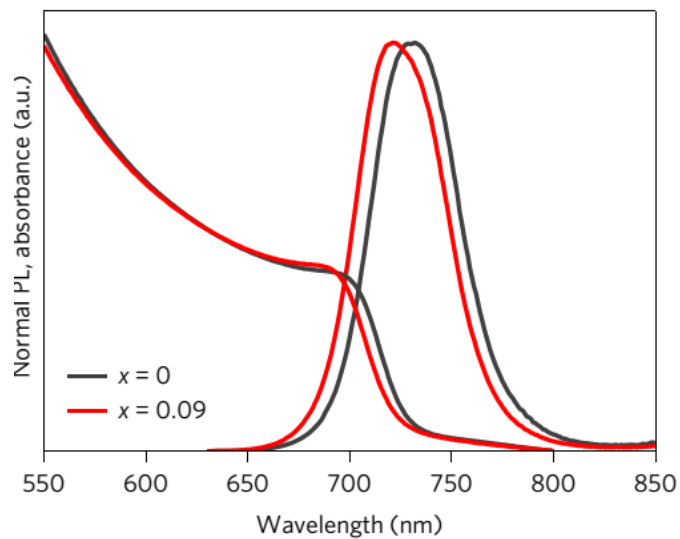
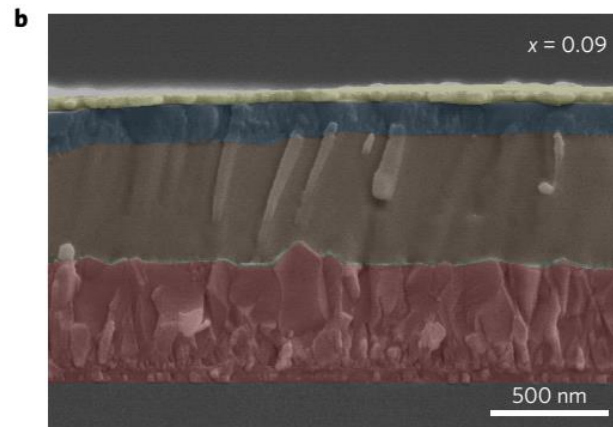


# 2D-3D structures

3D



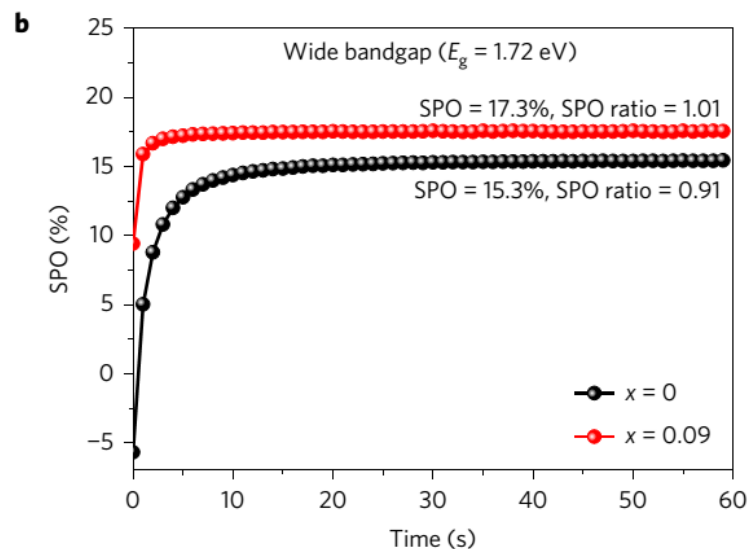
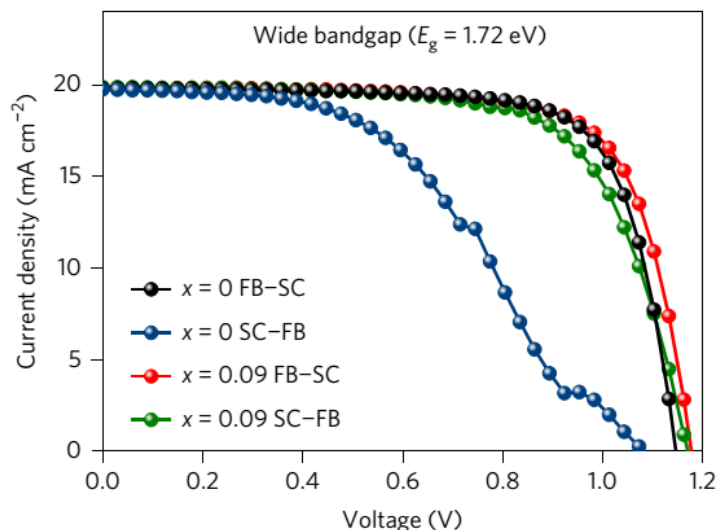
3D-2D







# 2D-3D structures

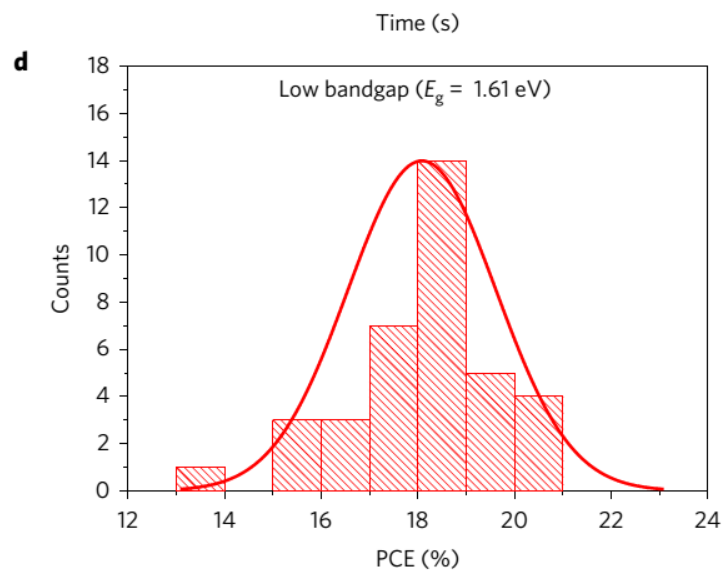
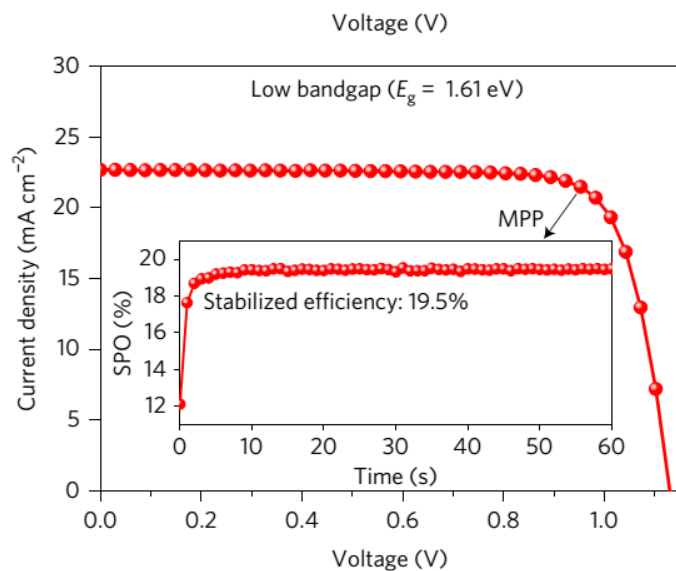


(a) J-V characteristics: 3D perovskite  $\text{FA}_{0.83}\text{Cs}_{0.17}\text{Pb}(\text{I}_{0.6}\text{Br}_{0.4})_3$  ( $x=0$ ) ( $E_g = 1.72$  eV) and for 3D-2D  $\text{BA}_{0.09}(\text{FA}_{0.83}\text{Cs}_{0.17})_{0.91}\text{Pb}(\text{I}_{0.6}\text{Br}_{0.4})_3$  ( $x=0,09$ )

(a) Stabilized cell efficiency (SPO) of the best cell (SPO ratio - ratio of SPO to PCE).



# 2D-3D structures



J-V characteristics: 3D perovskite for  $\text{BA}_{0.05}(\text{FA}_{0.83}\text{Cs}_{0.17})_{0.95}\text{Pb}(\text{I}_{0.8}\text{Br}_{0.2})_3$  ( $E_g = 1.61$  eV). Statistical distribution



# 2D-3D structures

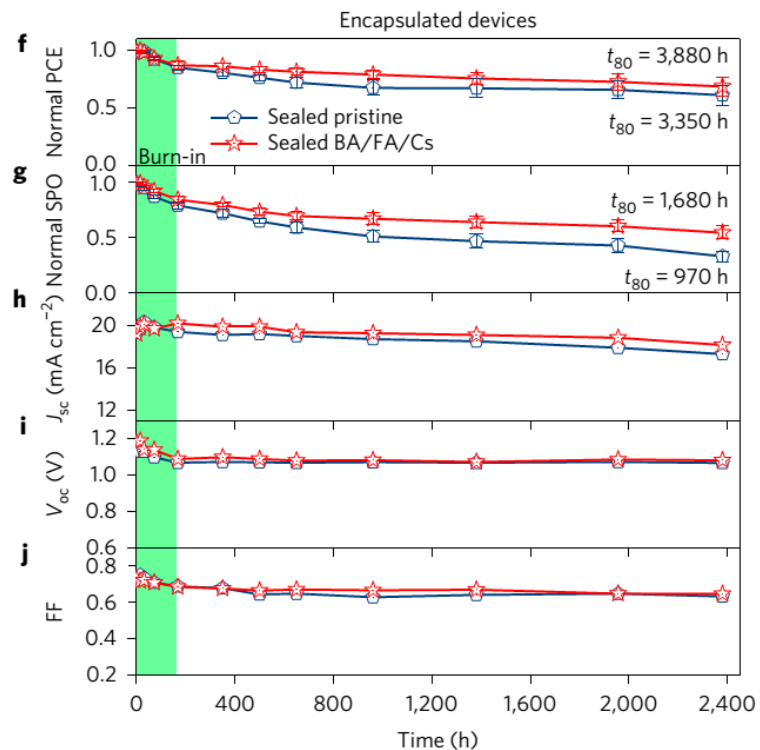
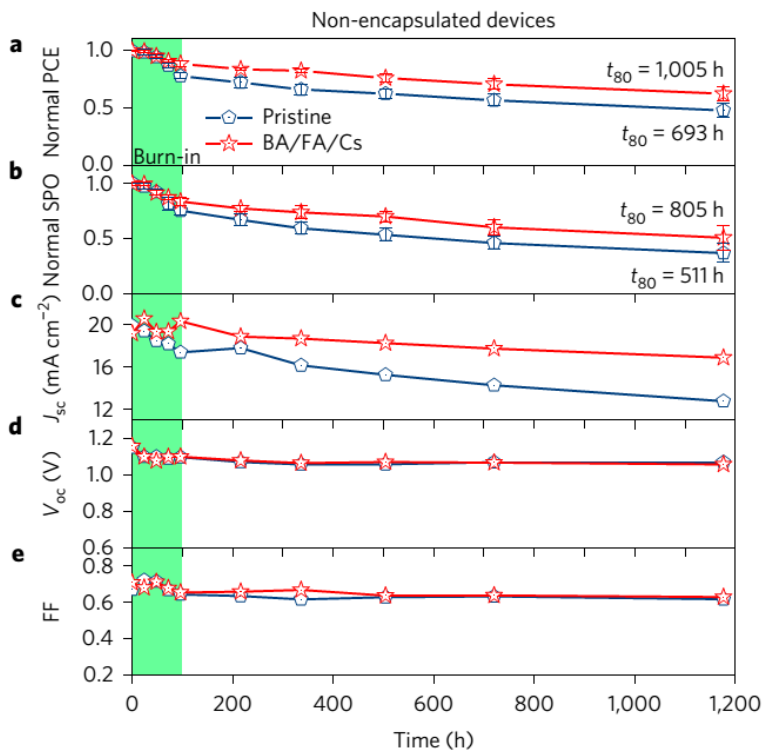
**Table 1 | Solar cell performance parameters determined from  $J$ - $V$  curves and stabilized power output measurements.**

Device	PCE (%)	$J_{sc}$ ( $\text{mA cm}^{-2}$ )	$V_{oc}$ (V)	FF	SPO (%)
<b>Wide-bandgap <math>\text{FA}_{0.83}\text{Cs}_{0.17}\text{Pb}(\text{I}_{0.6}\text{Br}_{0.4})_3</math> (<math>x = 0</math>)</b>					
Average	$15.1 \pm 1.0$	$18.8 \pm 0.9$	$1.15 \pm 0.02$	$0.70 \pm 0.03$	$14.1 \pm 0.9$
Champion	16.9	19.8	1.14	0.75	15.3
<b>Wide-bandgap <math>\text{BA}_{0.09}(\text{FA}_{0.83}\text{Cs}_{0.17})_{0.91}\text{Pb}(\text{I}_{0.6}\text{Br}_{0.4})_3</math> (<math>x = 0.09</math>)</b>					
Average	$15.5 \pm 1.1$	$18.9 \pm 0.7$	$1.17 \pm 0.02$	$0.70 \pm 0.04$	$15.8 \pm 0.8$
Champion	17.2	19.8	1.18	0.73	17.3
<b>Low-bandgap <math>\text{BA}_{0.05}(\text{FA}_{0.83}\text{Cs}_{0.17})_{0.91}\text{Pb}(\text{I}_{0.8}\text{Br}_{0.2})_3</math></b>					
Average	$18.1 \pm 1.5$	$22.1 \pm 0.7$	$1.09 \pm 0.04$	$0.75 \pm 0.05$	$17.5 \pm 1.3$
Champion	20.6	22.7	1.14	0.80	19.5

Average device characteristics with standard deviation were obtained on the basis of 32 cells for each set. The champion cell data are taken from the  $J$ - $V$  curves shown in Fig. 4.



# 2D-3D structures



Aging - AM1.5 xenon lamp with a power of  $76 \text{ mW/cm}^2$  in the air (approx. 45 RH%) without UV filter, in  $V_{oc}$  conditions, tested for different time intervals by a separate AM1.5 simulator with a power of  $100 \text{ mWcm}^{-2}$ . Light pulse aging with Suntest XLS+. The structure of the cell glass/FTO/SnO<sub>2</sub>/C60/perovskit/spiro-OMeTAD (with Li-TFSI and tBP) / Au.



## Selective growth of layered perovskite for stable and efficient photovoltaics.

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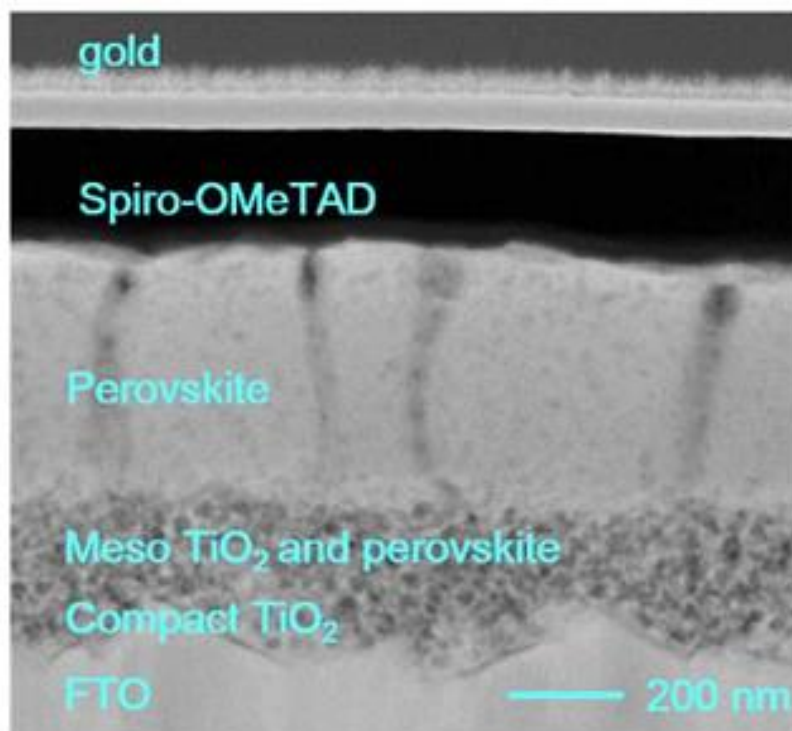
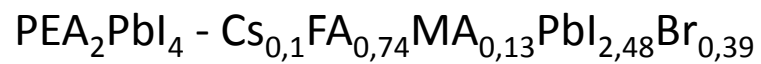
<sup>f</sup> Department of Energy Science, Sungkyunkwan University, 2066, Seobu-ro, Jangan-gu, Suwon 16419, Korea.

<sup>g</sup> Samsung Advanced Institute of Technology, 130, Samsung-ro, Yeongtong-gu, Suwon, 16678, Korea

<sup>h</sup> Business Support Department, Gumi Electrons & Information Technology Research Institute, Gumi, 39171, Korea



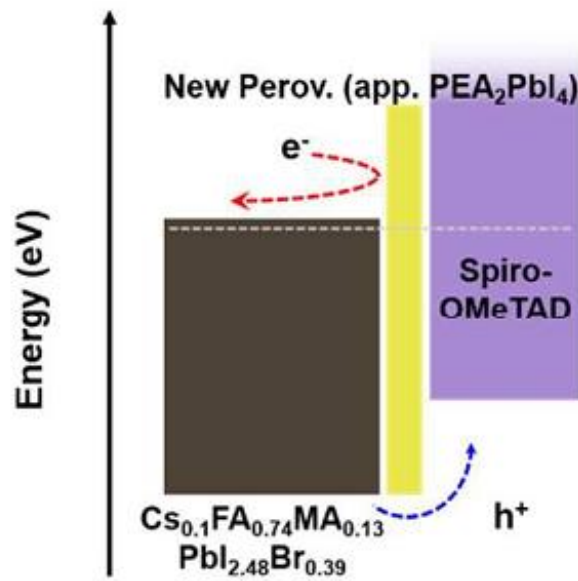
## 2D-3D structures



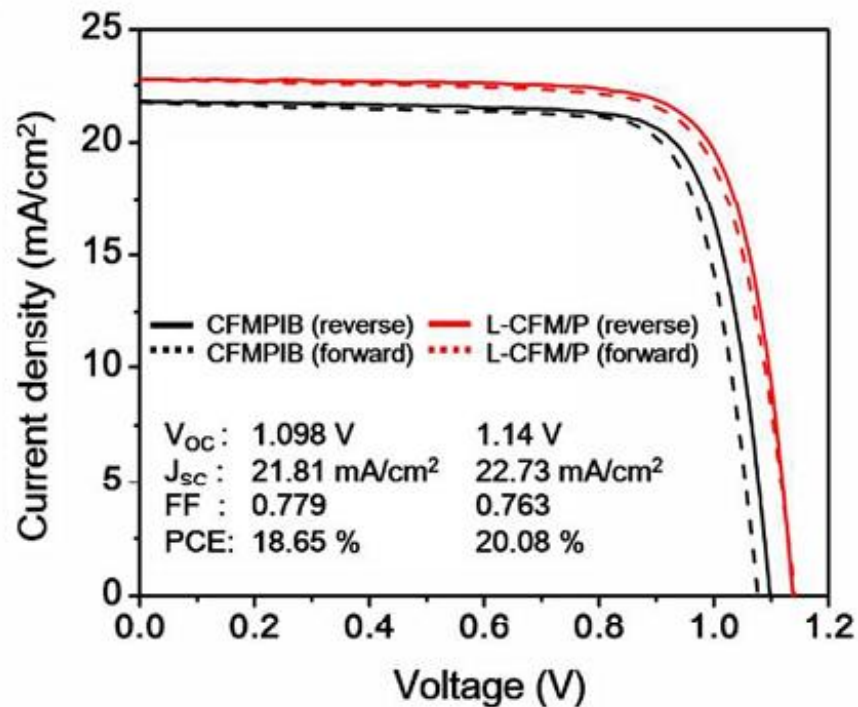


## 2D-3D structures

B



C



CFMPIB -  $\text{Cs}_{0.1}\text{FA}_{0.74}\text{MA}_{0.13}\text{PbI}_{2.48}\text{Br}_{0.39}$   
L-CFM/P (CFMPIB i  $\text{PEA}_2\text{PbI}_4$ )



# 2D-3D structures

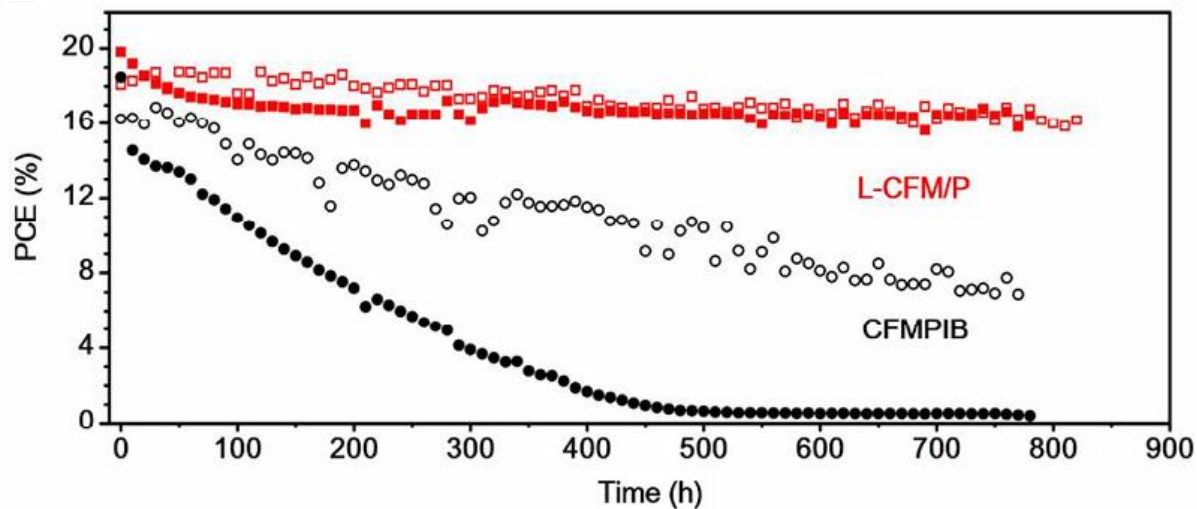
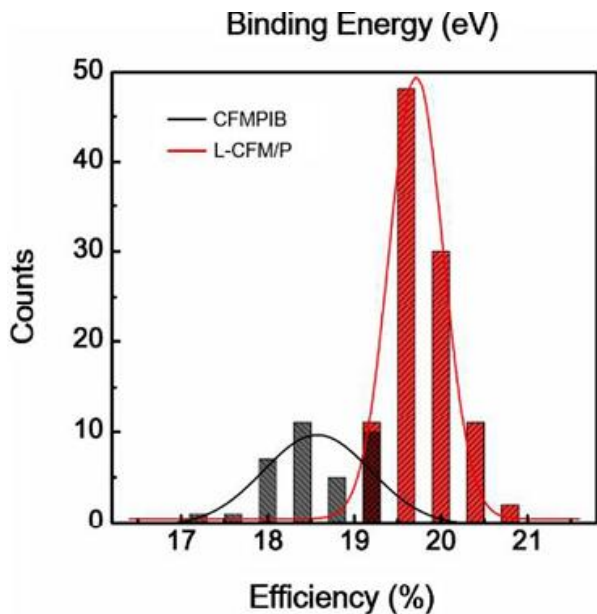
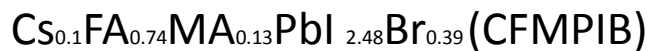


Photo-stability test for continuous (full) lighting in an inert atmosphere (blank stamps) and in the air (full) encapsulated under glass



L-CFM/P (perowskit CFMPIB i  $\text{PEA}_2\text{PbI}_4$ ).





## Recent progress in stability of perovskite solar cells\*

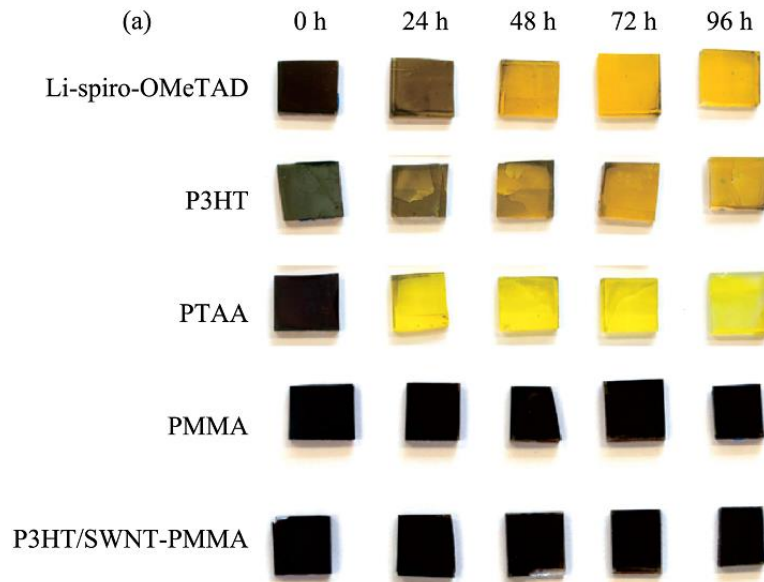
Xiaojun Qin<sup>1,2</sup>, Zhiguo Zhao<sup>1,2</sup>, Yidan Wang<sup>1,2</sup>, Junbo Wu<sup>1,2</sup>, Qi Jiang<sup>3</sup>, and Jingbi You<sup>3,4,†</sup>

<sup>1</sup>China Huaneng Group, Beijing 100031, China

<sup>2</sup>China Huaneng Clean Energy Research Institute, Beijing 102209, China

<sup>3</sup>Key Lab of Semiconductor Materials Science, Institute of Semiconductors, Chinese Academy of Sciences, Beijing 100083, China

<sup>4</sup>College of Materials Science and Opto-Electronic Technology, University of Chinese Academy of Sciences, Beijing 100049, China



PMMA- Poly(methyl 2-methylpropenoate)

P3HT- Poly(3-hexylthiophene)

PTAA -poly(triarylamine)

SWNT - carbon nanotube, single-walled



## ARTICLES

<https://doi.org/10.1038/s41560-017-0067-y>

nature  
energy

# Tailored interfaces of unencapsulated perovskite solar cells for >1,000 hour operational stability

Jeffrey A. Christians <sup>1</sup>, Philip Schulz <sup>1</sup>, Jonathan S. Tinkham<sup>2</sup>, Tracy H. Schloemer <sup>2</sup>,  
Steven P. Harvey <sup>1</sup>, Bertrand J. Tremolet de Villers <sup>1</sup>, Alan Sellinger <sup>1,2</sup>, Joseph J. Berry <sup>1\*</sup>  
and Joseph M. Luther <sup>1\*</sup>

<sup>1</sup>National Renewable Energy Laboratory, Golden, CO, USA. <sup>2</sup>Department of Chemistry, Colorado School of Mines, Golden, CO, USA.

NATURE ENERGY | VOL 3 | JANUARY 2018 | 68-74 |

## NREL Scientists Demonstrate Remarkable Stability in Perovskite Solar Cells

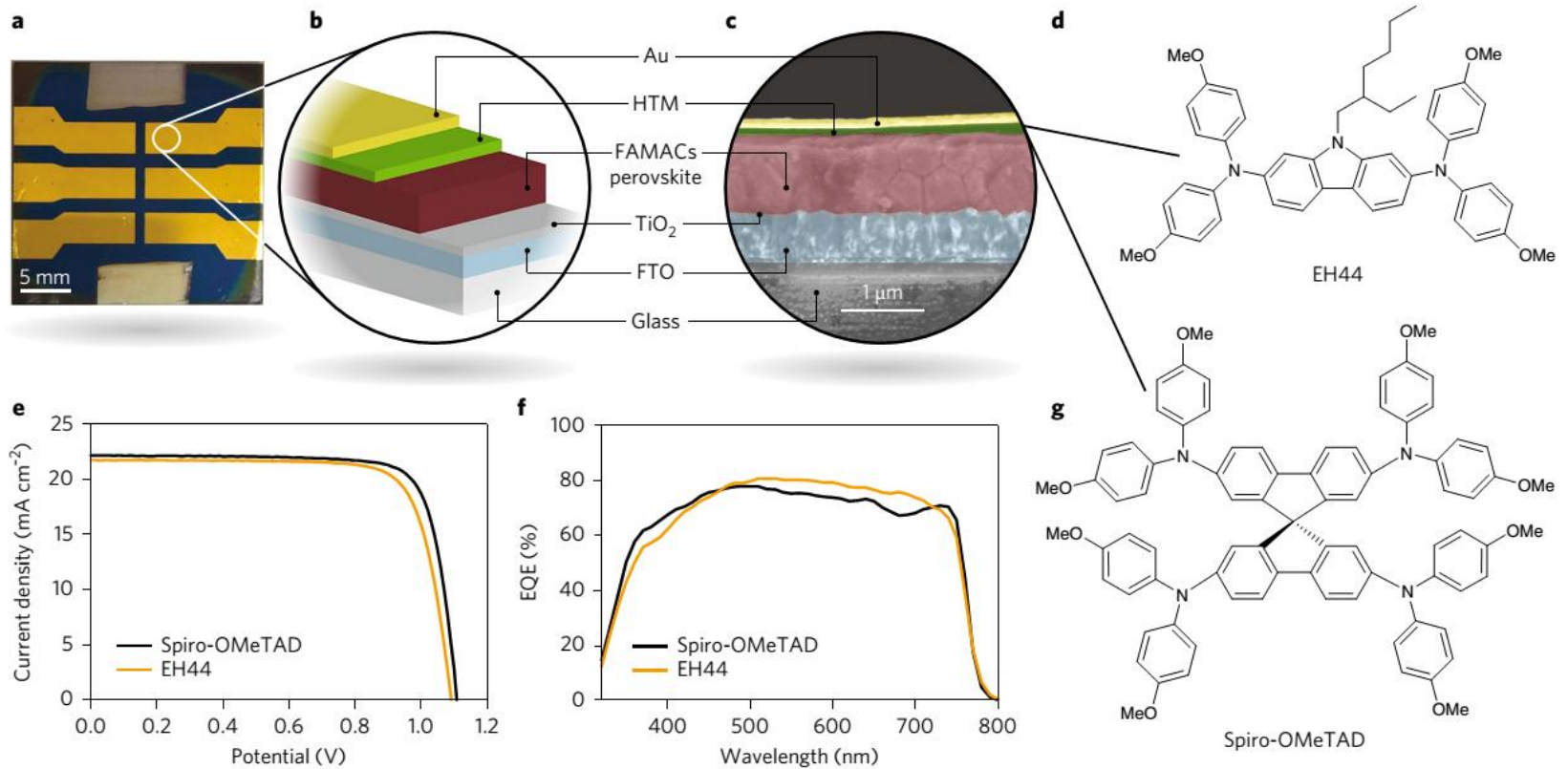
January 30, 2018

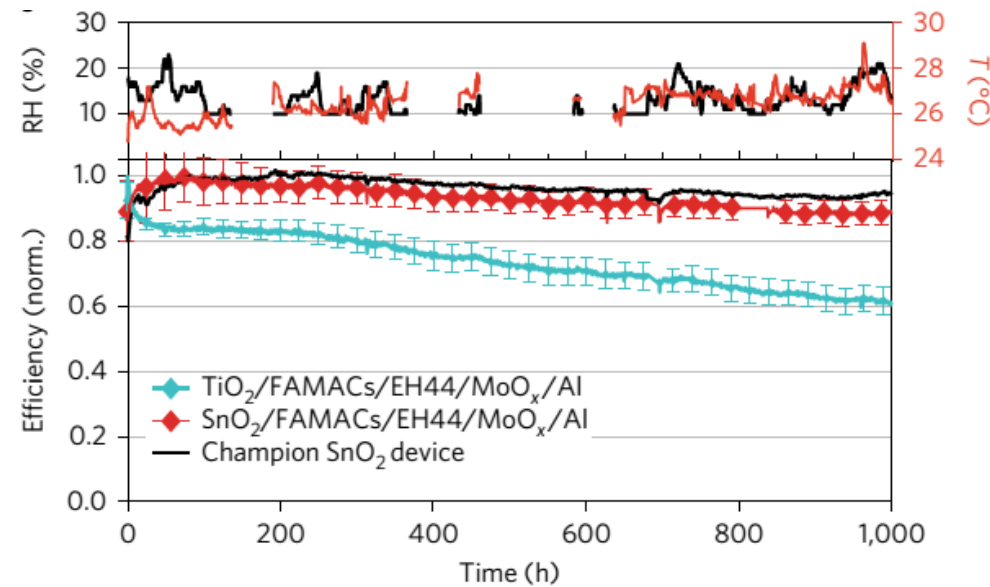
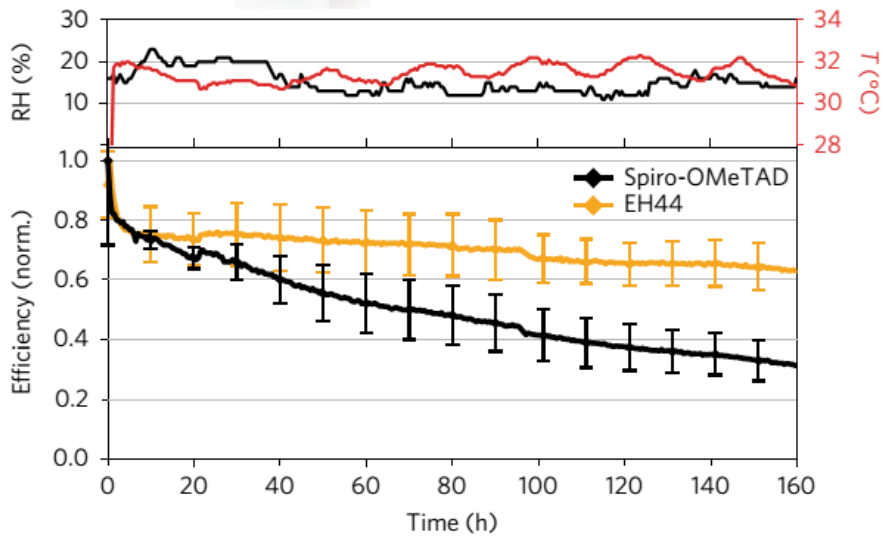
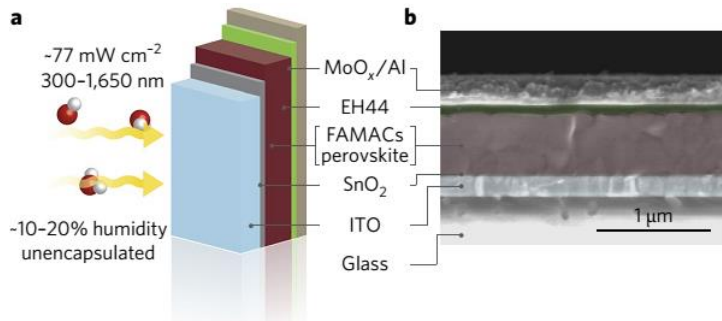
[https://www.nrel.gov/news/press/2018/nrel\\_scientists\\_demonstrate\\_remarkable\\_stability\\_in\\_perovskite.html](https://www.nrel.gov/news/press/2018/nrel_scientists_demonstrate_remarkable_stability_in_perovskite.html)



# New HTL and electrodes

NATURE ENERGY | VOL 3 | JANUARY 2018 | 68-74 |





Stability during operation of the TiO<sub>2</sub> / FAMACs / EH44 / Au (a) and ETL / FAMACs / EH44 / Mox / Al cells (ETL = TiO<sub>2</sub> (4 cells) or SnO<sub>2</sub> - 15 cells) (b) in air under certain conditions of humidity and temperature .

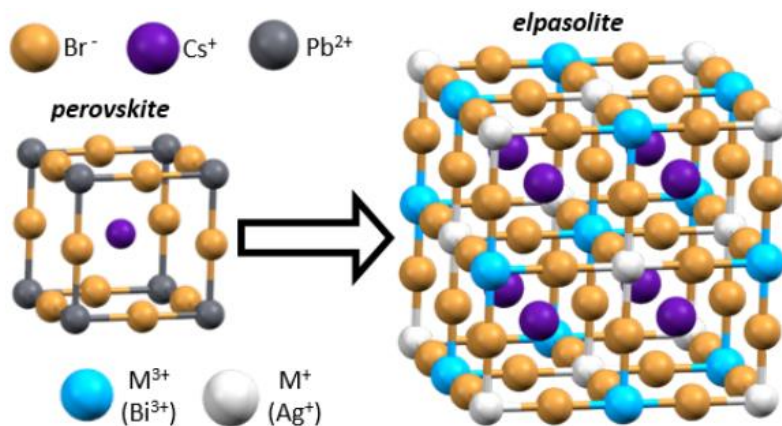


# Double halide perovskites

## Colloidal nanocrystals of lead-free double-perovskite (elpasolite) semiconductors: synthesis and anion exchange to access new materials

Sidney E. Creutz, Evan N. Crites, Michael C. De Siena, Daniel R. Gamelin\*

*Department of Chemistry, University of Washington, Seattle, WA 98195-1700, United States*



$\text{Cs}_2\text{AgBiBr}_6$  – doubling of the unit cell size and replacement of  $\text{Pb}^{2+}$  by  $\text{M}^+$  i  $\text{M}^{3+}$  cations

$\text{Cs}_2\text{AgInCl}_6$ ,  $\text{MA}_2\text{AgSbI}_6$ ,  $\text{MA}_2\text{TlBiBr}_6$ ,  $\text{MA}_2\text{KBiCl}_6$ , .....



## Earth-Abundant Nontoxic Titanium(IV)-based Vacancy-Ordered Double Perovskite Halides with Tunable 1.0 to 1.8 eV Bandgaps for Photovoltaic Applications

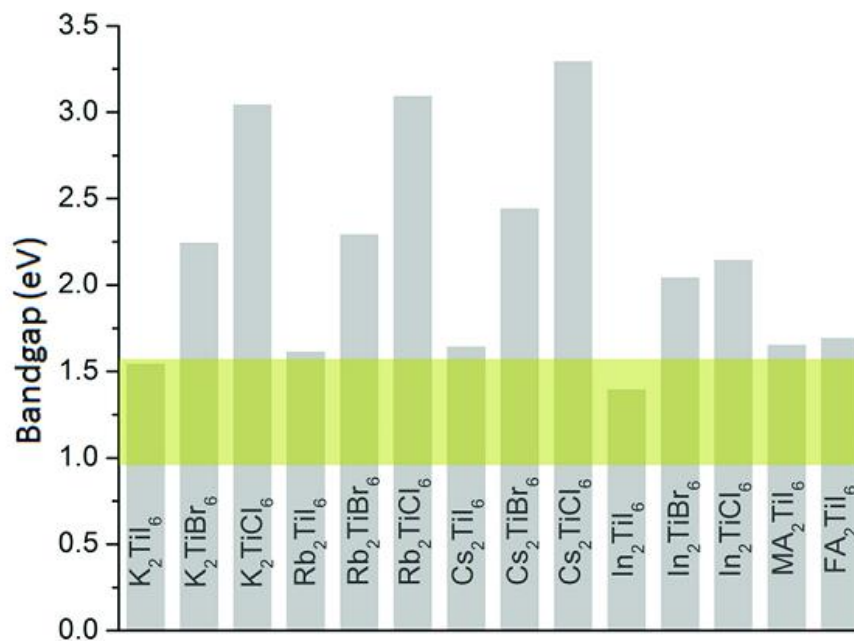
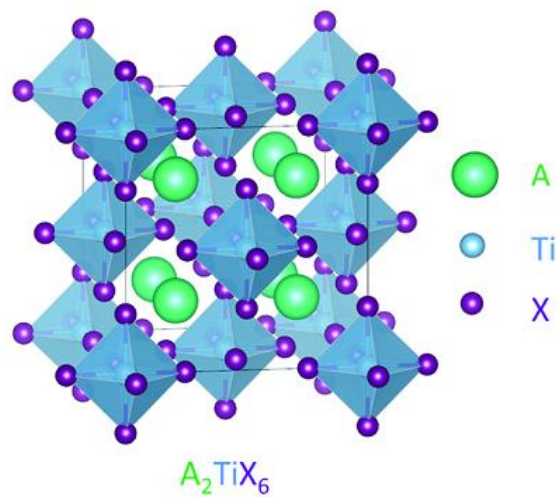
Ming-Gang Ju,<sup>†,§</sup> Min Chen,<sup>‡,§</sup> Yuanyuan Zhou,<sup>\*,‡</sup> Hector F. Garces,<sup>‡</sup> Jun Dai,<sup>†</sup> Liang Ma,<sup>†</sup>  
Nitin P. Padture,<sup>\*,‡</sup> and Xiao Cheng Zeng<sup>\*,†</sup>

<sup>†</sup>Department of Chemistry, University of Nebraska-Lincoln, Lincoln, Nebraska 68588, United States

<sup>‡</sup>School of Engineering, Brown University, Providence, Rhode Island 02912, United States



# Double halide perovskites





# Double halide perovskites

Journal of Materials Chemistry A

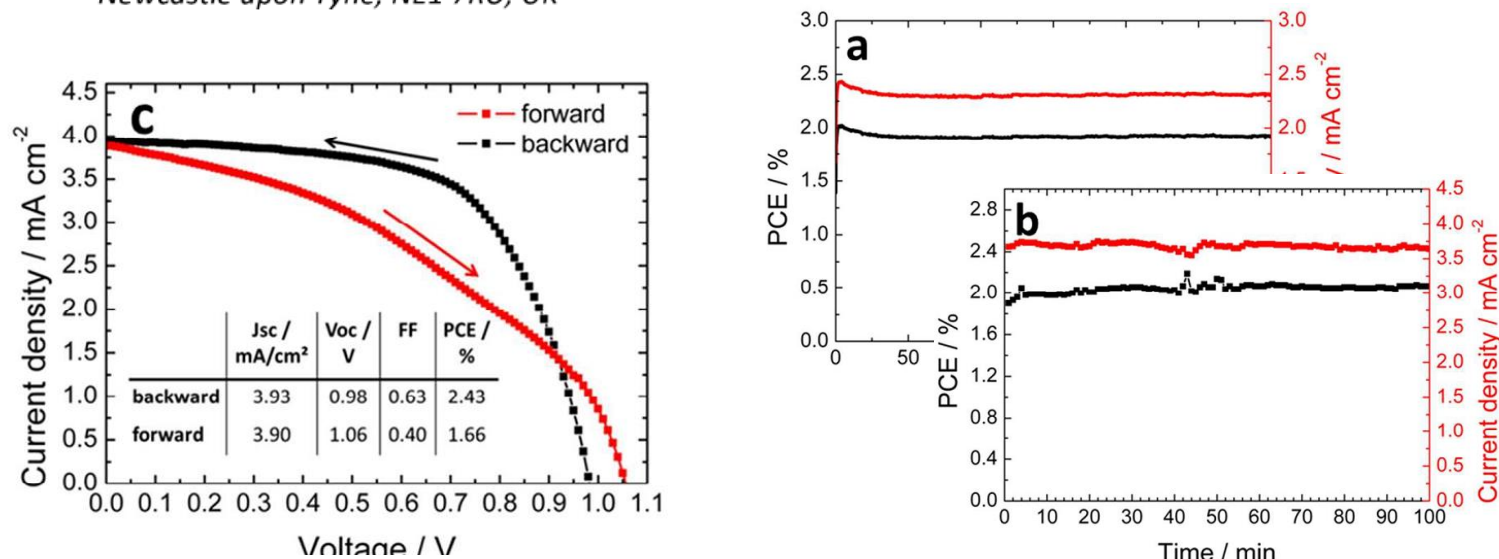
Published on 05 September 2017

## Highly stable, phase pure Cs<sub>2</sub>AgBiBr<sub>6</sub> double perovskite thin films for optoelectronic applications

Enrico Greul,<sup>a</sup> Michiel L. Petrus,<sup>a</sup> Andreas Binek<sup>a</sup>, Pablo Docampo<sup>b</sup> and Thomas Bein<sup>a\*</sup>

<sup>a</sup> Department of Chemistry and Center for NanoScience (CeNS), University of Munich (LMU) Butenandtstr. 5-13, 81377 Munich, Germany. \*E-mail: tbein@cup.uni-muenchen.de

<sup>b</sup> School of Electrical and Electronic Engineering, Newcastle University, Merz Court, Newcastle upon Tyne, NE1 7RU, UK



**Fig. 6** (a) Stabilized power output and current density measured under ambient conditions without encapsulation. (b) Photovoltaic performance as a function of time under continuous illumination under ambient conditions. All devices were manufactured according the procedure described in Fig. 1 with a 285 °C annealing step.



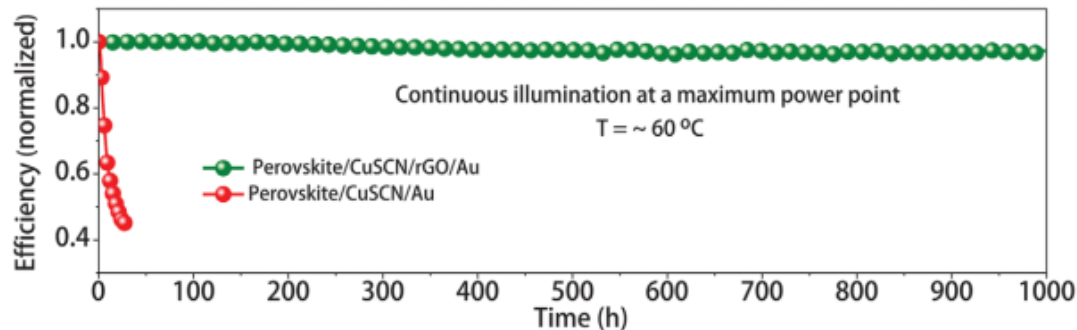


## SOLAR CELLS

*Science* **358**, 768–771 (2017) 10 November 2017

# Perovskite solar cells with CuSCN hole extraction layers yield stabilized efficiencies greater than 20%

Neha Arora,<sup>1\*</sup> M. Ibrahim Dar,<sup>1\*†</sup> Alexander Hinderhofer,<sup>2</sup> Norman Pellet,<sup>1</sup> Frank Schreiber,<sup>2</sup> Shaik Mohammed Zakeeruddin,<sup>1</sup> Michael Grätzel<sup>1†</sup>



The ways of increasing stability

CuSCN copper(I) thiocyanate

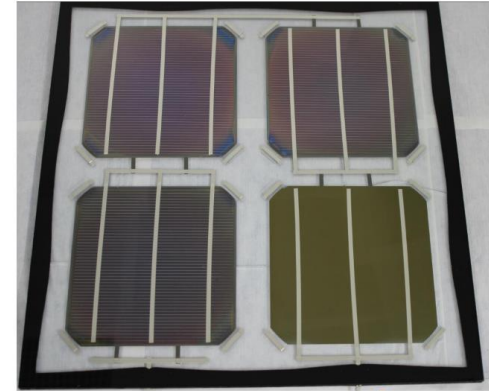
# Commercialization

**More than a 12 firms are involved in commercializing perovskite solar cells:**

- Energy Materials Corp.(US),
- Frontier Energy Solution (South Korea),
- Microquanta Semiconductor (China),
- Oxford PV (UK),
- Saule Technologies (Poland),
- Sekisui/Panasonic/Toshiba (Japan),
- Solaronix SA (Switzerland),
- Solliance (Netherlands), Swift Solar (US),
- Tandem PV (US),
- WonderSolar (China).



# Commercialization



 OXFORD PV

Oxford PV's industrial site in Brandenburg an der Havel, Germany, where the complete 250 MW production line will commence perovskite-on-silicon tandem solar cell production at the end of 2020.

## **Oxford PV tandem perovskite on the silicon pass the IEC 61646 test stability:**

200 thermal cycles (-40° C to +85° C) with <5 % drop, full sun light soaking 1000 hours (85%RH/85° C) with <4% drop, damp heat 1000 hours <4% drop)



The future of perovskite photovoltaic is bright. Perovskite solar cell technology is close to commercialization.

In the last few years there has been huge progress in the efficiency and in improving the stability of perovskite cells.

The perovskite /Si tandem cells has the greatest prospects for large-scale electricity production in near future.

Thank you for your attention